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Engineering two-dimensional superconductivity and Rashba spin-orbit coupling in $\text{LaAlO}_3/\text{SrTiO}_3$ quantum wells by selective orbital occupancy

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The discovery of two-dimensional electron gases (2DEGs) at oxide interfaces—involving electrons in narrow *d*-bands—has broken new ground, enabling the access to correlated states that are unreachable in conventional semiconductors based on *s*- and *p*- electrons. There is a growing consensus that emerging properties at these novel quantum wells—such as 2D superconductivity and magnetism—are intimately connected to specific orbital symmetries in the 2DEG sub-band structure. Here we show that crystal orientation allows selective orbital occupancy, disclosing unprecedented ways to tailor the 2DEG properties. By carrying out electrostatic gating experiments in $\text{LaAlO}_3/\text{SrTiO}_3$ wells of different crystal orientations, we show that the spatial extension and anisotropy of the 2D superconductivity and the Rashba spin-orbit field can be largely modulated by controlling the 2DEG sub-band filling. Such an orientational tuning expands the possibilities for electronic engineering of 2DEGs at $\text{LaAlO}_3/\text{SrTiO}_3$ interfaces.

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The confinement of electron orbitals over small scales provides a pathway to tailor the electronic properties of a quantum system. Restricting the motion of electrons within planes of different crystal orientation affords additional routes to reorganize the electronic band structure. Such strategies have been used to engineer the electronic and optical properties of II–VI and III–V semiconductor quantum wells, where *s*- and *p*- orbitals are involved^{1–3}. Yet, two-dimensional (2D) electron gases (2DEGs), comprising *d*-electrons instead of *s* or *p*, have come into the limelight over about the last 10 years^{4–11}, opening novel perspectives that are inaccessible for more traditional materials. In particular, the narrow bandwidth of *d*-states in transition metal oxide quantum wells promote correlated states—for example, magnetism and superconductivity—that are unseen in conventional semiconductors. Interestingly, the extreme confinement, over just a few unit cells, enables full electrostatic control of correlated 2DEG states, allowing access to new physics and paving the way to new device concepts. Particularly, these 2D electron systems have been found to be superconductive^{7,12–14} with 2D superconductivity largely modulated by electric gates^{15,16}. The 2D character of the superconductivity has led to phenomena not observed in the 3D regime, such as magnetic enhancement of superconductivity¹⁷, violation of the paramagnetic Pauli limit for the upper critical fields¹⁸, quantum phase transitions¹⁹ or multiple quantum criticality²⁰. The intricacy of all these complex phases and the evidence of the role of electron correlations have often prompted the use of the concept of electron liquids to designate these electron systems²¹.

The interface between LaAlO₃ and SrTiO₃ is the oxide quantum well par excellence. Initially, the research on LaAlO₃/SrTiO₃ quantum wells was restricted to the (001)-plane of the perovskite unit cell^{4,14,15}. Remarkably, recent investigations have uncovered that interface conductivity also appears along other directions, such as (110) (refs. 22,23) and (111) (ref. 22). The selective confinement of electrons within planes of different crystal orientation expands vigorously the possibility of fine-tuning the 2DEG sub-band hierarchy and, thereof, the physical properties. Along this line, we have recently demonstrated that crystal symmetry is an extra degree of freedom to realize different 2DEG band reconstructions at the LaAlO₃/SrTiO₃ interface, by imposing distinctive orbital hierarchies on (001)- and (110)-oriented quantum wells and enabling the selective occupancy of states of different symmetry²⁴. More specifically, we have uncovered that the degeneracy within the *t*_{2g} sub-band—which forms the backbone of the 2DEG structure in LaAlO₃/SrTiO₃ wells—is broken in reversed ways depending on the crystal orientation: for (001)-oriented 2DEGs the *d*_{xy} orbitals have the lowest energy, while along (110) the bottommost levels have instead a *d*_{xz}/*d*_{yz} character²⁴. Recent experiments on uncapped (110) SrTiO₃ surfaces also found the same hierarchy²⁵. This orbital reconfiguration provides an excellent playground to test the link between orbital symmetry and complex correlated states, provided that we understand exactly the implications that such 2DEG band engineering has for the physical properties of the quantum wells.

In this work, we present evidence that the selection of the orbital symmetries in the 2DEG sub-band structure triggers some nontrivial and extensive modifications of the electronic properties of quantum wells at the LaAlO₃/SrTiO₃ interface. First, we demonstrate that the orbital reconfiguration implies a modulation of 2DEG spatial extension and, as a result, the anisotropy of the 2D superconductivity is largely affected by crystal orientation. Second, we show that the effects of sub-band engineering are influential on the spin–orbit coupling and the concomitant Rashba effect, opening new pathways to tune the spin–dependent transport in LaAlO₃/SrTiO₃ quantum wells. These findings open

fresh perspectives to understand the fundamental connection between orbital symmetry and the electronic phases at LaAlO₃/SrTiO₃ interfaces.

Results

Structural characterization. The samples analysed here were obtained by pulsed laser deposition of LaAlO₃ thin films on TiO₂-terminated (001)-SrTiO₃ substrates (LaAlO₃ thickness *t* = 10 monolayers (MLs), corresponding to *t* ~ 3.8 nm) as well as on thermally treated (110)-oriented SrTiO₃ substrates (*t* = 7–14 MLs, *t* ~ 1.9–3.8 nm), see details in Methods and (refs 22,26,27). We carried out cross-sectional scanning transmission electron microscopy (STEM) in the high-angle annular dark field (HAADF) imaging mode, in which, to a good approximation, the intensity of an atomic column is proportional to the square of the atomic number (*Z*), so elements can be deduced by tracking column intensities²⁸. Brighter atomic columns correspond to the heavier elements, La and Sr, whereas fainter columns correspond to Ti and Al. Atomic-scale structural characterization shows a coherent and epitaxial growth of both heterostructures and atomically flat interfaces—Fig. 1a,b for (001) and (110), respectively— Besides, regarding the (110)-oriented sample, along the [001] zone axis the (110) ionic stacking across the interface can be readily appreciated, see Fig. 1b. Therefore, in spite of the higher surface energies of (110)-planes with respect to (001), the STEM-HAADF study rules out altogether the formation of (100) microfacets at the (110)-interface^{23,29,30}.

Spatial extension and anisotropy of 2D superconductivity.

We discuss first the implications of band reconstruction on the 2DEG superconductivity. In line with previous reports on (001) (refs 14,15,19), we show that the (110)-interface is also superconductive and has a 2D character. Yet, we uncover that the anisotropy of the 2D superconductive state is considerably larger for (001) than for (110). Such a conclusion is readily apparent from the sheet resistance curves measured under the magnetic fields applied in-plane (Fig. 2a,b). It is known that as the 2D limit is approached, increasingly higher in-plane fields are required to suppress the superconductivity, since vortex entry is impeded by the low dimensionality¹³. Therefore, higher in-plane critical fields imply stronger anisotropy. Inspection of Fig. 2a,b shows that the (001) interface requires much higher in-plane fields ($\mu_0 H_{c2,\parallel} \approx 2,200$ mT) than the (110) interface ($\mu_0 H_{c2,\parallel} \approx 1,000$ mT) to induce the transition to the normal state. We conclude, thus, that the 2D anisotropy is larger for (001) than for (110), anticipating a smaller spatial extension of the quantum well along (001).

For a quantitative estimation of both the superconductive layer thickness *d* and the in-plane superconductive coherence length ξ , we carried out an analysis based on the Landau–Ginzburg formalism³¹. For that purpose, the out-of-plane $\mu_0 H_{c2,\perp}$ and in-plane $\mu_0 H_{c2,\parallel}$ critical fields were determined by defining quantitative criteria for the field-induced transitions. Thus, a drop resistance of 90% from the normal resistance state at *T* = 400 mK was established to ascertain the evolution of the transition temperature *T*_C. We consider first the (110) sample with LaAlO₃ thickness *t* = 14 MLs. The out-of-plane critical field, extrapolated to *T* = 0 K, was $\mu_0 H_{c2,\perp} \approx 160$ mT (Supplementary Fig. 1), leading to an in-plane coherence length $\xi = \sqrt{\frac{\Phi_0}{2\pi\mu_0 H_{c2,\perp}}} \approx 44$ nm (Φ_0 is the flux quantum)³¹. In addition, from the in-plane critical field $\mu_0 H_{c2,\parallel} \approx 1,000$ mT we could estimate the superconductive thickness $d = \frac{\Phi_0\sqrt{3}}{\pi\mu_0 H_{c2,\parallel}\xi} \approx 24$ nm. Since the coherence length is well above the superconductive thickness (*d* < ξ), the superconductivity is shown to be 2D. Applying the

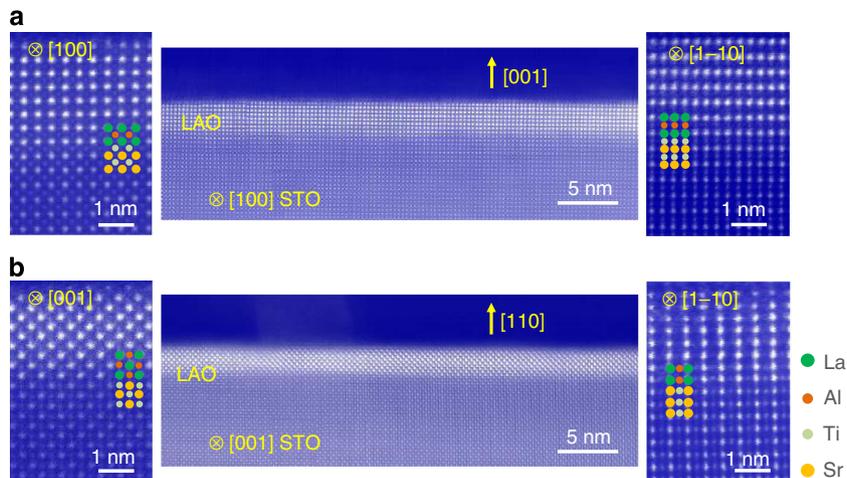


Figure 1 | Atomic-resolution STEM characterization. (a) HAADF-STEM images of the LaAlO₃/SrTiO₃ (001) interface. The left and right panels are magnified views of the interface observed from [100] and [1-10] zone axes, respectively. (b) HAADF-STEM images of the LaAlO₃/SrTiO₃ (110) interface. Left and right panels are magnified views of the interface observed from [001] and [1-10] directions, respectively. Both LaAlO₃ layers are continuous within the analysed region (of the order of 1 μm). The images in the central panels **a** and **b** have been Fourier filtered to reduce background noise. The positions of La and Sr are indicated by green and orange circles, whereas Al and Ti are shown in red and light green. Note that for both orientations the interfaces are atomically flat and that the (110) interface does not show any local (100) microfacet.

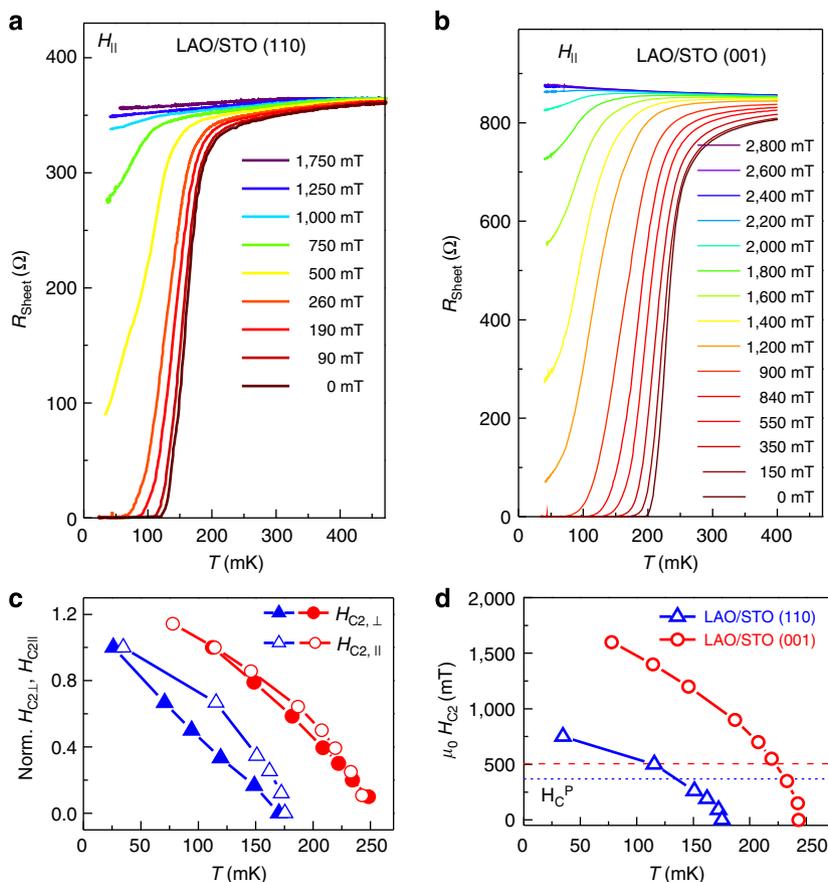


Figure 2 | Anisotropy of the 2D superconductivity at the (001) and (110) interfaces. Sheet resistance of (a) the (110)-interface with $t = 14$ MLs and (b) the (001)-interface with $t = 10$ MLs, under magnetic fields applied parallel to the interface. The field values are indicated in the panels. Panel (c) shows the temperature dependence of out-of-plane $\mu_0 H_{c2,\perp}$ and in-plane $\mu_0 H_{c2,\parallel}$ critical fields of (001)—red circles—and (110)—blue triangles—interfaces, corroborating the 2D character of the superconductivity for both orientations. (d) The upper critical fields are displayed as a function of the temperature for both the orientations (field in-plane). The dotted and dashed straight lines indicate the Pauli-limited critical fields $\mu_0 H_C^P$. The observation of higher critical fields for the (001) interface is consistent with the larger anisotropy of the 2DEG superconductivity and stronger spatial confinement for (001).

same protocol analysis to the other (110)-interfaces in this study, with thickness in the range $t = 7\text{--}10$ MLs (Supplementary Fig. 2), we find that always the in-plane coherence length ($\xi \approx 40\text{--}75$ nm) is significantly larger than the superconductive thickness ($d \approx 24\text{--}30$ nm), thus confirming the 2D character of the superconductivity at the (110) interface. This is also corroborated by the analysis of the temperature dependence of the resistance, showing that the transition to the superconductivity at (110) interfaces belongs to the Berezinskii–Kosterlitz–Thouless (BKT) universality class^{32–34} (Supplementary Fig. 3). In addition, the out-of-plane and in-plane critical fields follow the temperature dependence expected for 2D superconductors (Fig. 2c), that is, $\mu_0 H_{c2,\perp} = \frac{\Phi_0}{2\pi\xi^2} (1 - T/T_C)$ and $\mu_0 H_{c2,\parallel} = \frac{\Phi_0\sqrt{12}}{2\pi\xi d} (1 - T/T_C)^{1/2}$, respectively¹³.

We applied also the Landau–Ginzburg analysis to a (001) LaAlO₃/SrTiO₃ sample using the same growth conditions as those used for the (110)-oriented samples. The analysis of the experimental data concludes that the coherence length is $\xi \approx 40$ nm and the superconducting thickness is $d \approx 13$ nm, in close agreement with the values previously reported^{14,35}. We, thus, demonstrate in a quantitative manner that the spatial extension of superconductive (110) interfaces ($d \approx 24\text{--}30$ nm) is considerably larger than the one usually reported for (001) interfaces ($d \approx 10\text{--}13$ nm) (refs 14,35).

The wider spatial extent of the (110)-2D state is also inferred from the analysis of the Pauli paramagnetic limit of the upper critical fields. For high-enough magnetic fields, the paramagnetic susceptibility induces a parallel alignment of the Cooper pair spins that eventually breaks them apart, giving a higher bound for the upper critical fields^{18,36}. This value can be assessed as $\mu_0 H_c^p \approx 1.76 k_B T_C / \sqrt{2} \mu_B$, where k_B is the Boltzmann's constant and μ_B is the Bohr magneton (assuming a g factor of 2)^{18,36}. Although this upper bound is generally fulfilled, it is violated in some cases. One example is the case of ultrathin SrTiO₃ 2D superconducting layers for which the values of $\mu_0 H_{c2,\parallel}$ were found to exceed largely the Pauli limit. This was explained by the large intrinsic spin–orbit coupling at interfaces, which becomes a prominent energy scale as the thickness is reduced¹⁸. The correlation between the spatial confinement and the anisotropy of the 2D superconductivity is also borne out in the (001) and (110) LaAlO₃/SrTiO₃ interfaces. Figure 2d summarizes this observation: the upper critical fields $\mu_0 H_{c2,\parallel}$ measured in (001) interfaces are significantly higher than those measured in (110) samples at any temperature. As a matter of fact, for the (001) interface the Pauli limit is already violated at temperatures below $T \leq 220$ mK, close to T_C . Instead, the Pauli limit is only surpassed at temperatures $T \leq 110$ mK for the (110) interface, further away from the transition (Fig. 2d). Again, this is an indication of stronger 2DEG confinement at the (001) interface.

Electrostatic modulation of 2D superconductivity. The different 2DEG spatial extent has also consequences on the electrostatic modulation of the superconductivity. We performed electrostatic gating experiments in (001)- and (110)-oriented samples that were contacted by top and backgate electrodes and electric fields were applied in the range of $V_g = \pm 400$ V (Fig. 3). Positive/negative voltages correspond to the accumulation/depletion of electrons at the interface, respectively. Hall and capacitance experiments allowed us to obtain the sheet carrier density modulation as a function of the voltage V_g for both the film orientations. The curves of carrier density that we extract from Hall measurements exhibit a reduction of n_{Hall} for positive V_g (Fig. 3f). Such a feature is the hallmark of multiband conduction, in which high- and low- mobility carriers participate in the transport in the regime of accumulation, whereas only one type of carrier is

relevant in the regime of depletion ($V_g \ll 0$) (ref. 16). The total carrier density n_s , comprising both heavy and light electron bands, can be obtained by experiments that measure the capacitance between the backgate and the 2DEG. In this case, the value of n_s is extracted by integration over the voltage range $n_s(V_g) = n_s(V_g = -V) + \frac{1}{eA} \int_{-V}^V C(V) dV$, where A is the area of the capacitor. Note that, in agreement with the two-carrier scenario, only one band is involved in transport at negative V_g and n_s is superimposed to n_{Hall} within this range of applied voltages (Fig. 3f). Instead, in the regime of accumulation, $V_g > 0$, two bands are involved and n_s and n_{Hall} differ significantly^{16,37}.

Figure 3e summarizes the results of the electrostatic gating experiments, where the superconducting transition temperature T_C and the resistance R_{sheet} at the normal state are plotted as a function of the gate voltage V_g . We see that the carrier density is largely modulated for both orientations, with variations $\Delta n_s = 0.2 - 0.8 \times 10^{14} \text{ cm}^{-2}$ and $\Delta n_s = 0.4 - 1.6 \times 10^{14} \text{ cm}^{-2}$ for (001) and (110) interfaces, respectively. However, despite similar modulations of the carrier density for both orientations, their effects on the superconductivity are dramatically different depending on the crystal orientation. More specifically, the superconductivity of the (001)-interface could be suppressed for a range of applied fields (Fig. 3a), in agreement with previous reports¹⁵. At the (001) interface the $T_C(V_g)$ curve exhibits a dome-like shape (Fig. 3e), indicating that superconductivity is suppressed at fields above $V_g \approx +200$ V and below $V_g \approx -50$ V. Instead, for (110) interfaces the superconducting state is never switched off by electric fields (Fig. 3b) and the transition temperature is modulated by at most about 50% (Fig. 3e). The much larger tunability of (001) interfaces with respect to (110) is again consistent with the narrower extension of 2DEGs at (001) wells.

Modulation of the Rashba spin–orbit field. Previous works have demonstrated that there is a strong spin–orbit field that stems from a Rashba-type interaction at the LaAlO₃/SrTiO₃ interface^{38,39}. As a result, an effective magnetic field B_{SO} is felt by electrons moving relativistically under the influence of the interface intensive electric fields $E_0 = -\nabla V(r)$. Remarkably, the intensity of B_{SO} is directly related to electron hopping between t_{2g} orbitals that, although forbidden in the unperturbed system away from interfaces, are however allowed in the presence of the field E_0 (ref. 40). In particular, E_0 induces a polarization of the atomic orbitals, which break their symmetry and, as a consequence, allows a hybridization within the t_{2g} manifold in the metal–oxygen network that contributes to B_{SO} (refs 40,41). Because of the different 2DEG band structure along (001) or (110), the spin–orbit field B_{SO} is expected to have a strong orientational dependence.

To probe the effects of orientational reconstruction on the spin–orbit term B_{SO} , we analysed the field dependence of the magnetoconductance at the normal state recorded at a temperature $T = 3.3$ K under applied electric fields (Fig. 4a,b). The experimental data were fitted to the expression^{38,42}

$$\frac{\Delta\sigma(B)}{G_0} = -\Psi\left(\frac{1}{2} + \frac{B_{\text{tr}}}{B}\right) + \frac{3}{2}\Psi\left(\frac{1}{2} + \frac{B_{\phi} + B_{\text{SO}}}{B}\right) - \frac{1}{2}\Psi\left(\frac{1}{2} + \frac{B_{\phi}}{B}\right) - \left[\ln\left(\frac{B_{\phi} + B_{\text{SO}}}{B_{\text{tr}}}\right) + \frac{1}{2} \ln\left(\frac{B_{\phi} + B_{\text{SO}}}{B_{\phi}}\right) \right] - A_K \frac{\sigma(0)}{G_0} \frac{B^2}{1 + CB^2} \quad (1)$$

that describes the change of conductivity with field $\Delta\sigma(B)$ normalized by the quantum of conductance $G_0 = e^2/\pi h$ (refs 38,42,43). In equation (1), quantum corrections to the conductance in the 2D limit are described by the four first terms, where $\Psi(x)$ is the digamma function, and B_{tr} , B_{ϕ} and B_{SO} are the

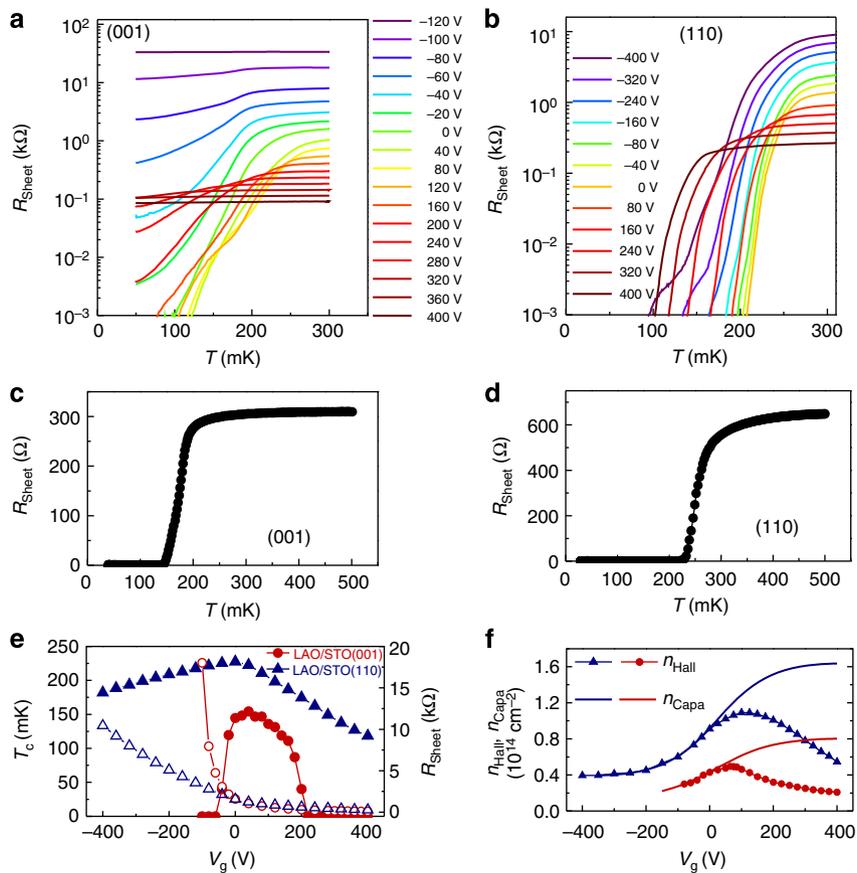


Figure 3 | Electrostatic modulation of the 2D-superconductivity at the (001) and (110) interfaces. Temperature dependence of the resistance as a function of gate voltage for (a) (001)—and (b) (110)-oriented LaAlO₃/SrTiO₃ interfaces, respectively. In panels (c) and (d) we show the superconducting transitions in linear scale, for zero applied voltage, for (001) and (110) interfaces, respectively. (e) Superconducting T_c (filled symbols) and sheet resistance (open symbols) of (001)—and (110)-interfaces plotted against the gate voltage. (f) Hall carrier density of (001)- (circles) and (110)- (triangles) interfaces. Solid lines correspond to the carrier density obtained from the analysis of the capacitance.

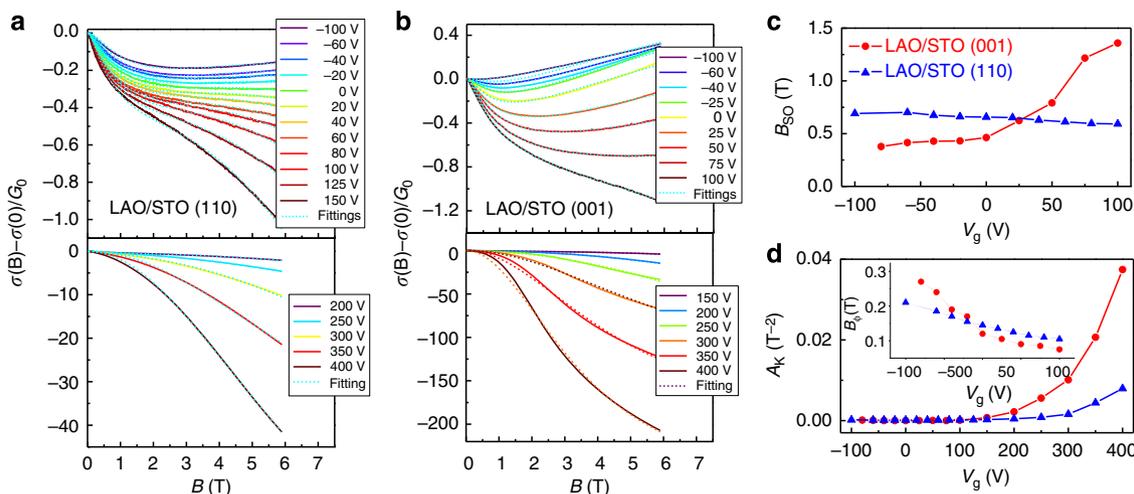


Figure 4 | Analysis of the spin-orbit coupling at the (001) and (110) interfaces. The field-dependent magnetoconductance $\Delta\sigma(B)$ normalized by the quantum of conductance G_0 was measured under different gate voltages V_g . Data measured at $T=3.3$ K are shown for the (a) (110) and (b) (001) interfaces. Dashed lines are the fittings to equation (1) and the values of the applied voltage are indicated in the panels. The dependence of the spin-orbit term B_{SO} on the gate voltage V_g is plotted (c). The Kohler A_K and inelastic B_ϕ terms obtained from fittings to equation (1) are plotted in (d) and inset, respectively.

effective fields related to the elastic, inelastic and spin-orbit scattering terms, respectively⁴³. Finally, the last term in equation 1, involving the parameters A_K and C , is the Kohler

term that gives an account of orbital magnetoresistance. Fittings of the experimental data to equation (1) were excellent, as shown in Fig. 4a,b for both orientations and for different electric fields.

The parameters B_{SO} , A_K and B_ϕ extracted from these fittings are shown in Fig. 4c,d. It turns out that the Kohler term A_K became rather large at positive fields $V_g > +100$ V (Fig. 4d), making difficult a precise evaluation of the spin-orbit term in the regime of strong electron accumulation. For that reason in Fig. 4c we plotted the evolution of the term B_{SO} restricted to the range $V_g = \pm 100$ V, where accurate values of the spin-orbit contribution can be obtained.

For the (001) sample, the values of B_{SO} that we obtained from fittings to equation (1) are in the same range as reported previously for the same orientation, with a similar asymmetric dependence of the spin-orbit field with V_g (refs 38,43). In the regime of depletion ($V_g < 0$ V), the values of the spin-orbit field are $B_{SO} < 0.5$ T, whereas for electron accumulation ($V_g > 0$ V) the spin-orbit term rises up to $B_{SO} \approx 1.5$ T. We thus observe a strong asymmetric field dependence of B_{SO} for the interfaces along (001). In contrast, the electrostatic modulation of B_{SO} is very weak along (110), and the spin-orbit field is largely unaffected by the electrostatic gating, with values restricted within a much narrower range $B_{SO} \approx 0.6$ – 0.7 T (Fig. 4c). In brief, our analysis demonstrates that the Rashba spin-orbit fields at the (110) interfaces are substantially different from those along (001). This observation illustrates how band engineering based on crystal symmetry can be exploited to tailor the spin-dependent transport along SrTiO₃-based quantum wells^{44,45}.

Discussion

The different spatial extension of the quantum wells along (001) and (110) and the different behaviour of the Rashba spin-orbit fields can be elucidated on the grounds of the modulation of the 2DEG sub-band structure observed in the experiments²⁴ that, in turn, can be understood using the fundamental concepts of quantum physics of solids. When we consider the orbitals of t_{2g}

electrons that are confined along (001) or (110), the quantum well entrapment of d_{xy} , d_{xz} and d_{yz} wavefunctions produces an energy splitting between the different eigenstates that is inversely proportional to their effective masses along the confinement direction⁴⁶. Figure 5 illustrates schematically the arguments that we expose in the following. Note that although the full complexity of the quantum sub-band structure^{47,48} is ignored in this Figure—as we depict only one sub-band for each type of orbital—the essential physics is captured. More specifically, for confinement along (001), π -type bonding between d_{xy} states leads to small wavefunction overlapping and large effective mass, while along (110) σ -like bonds between d_{xy} orbitals lead to much smaller effective mass (Fig. 5a,b). Instead, the overlapping of d_{xz}/d_{yz} states has intermediate values for both the orientations. This results in a hierarchy of out-of-plane effective masses given by $m_{xy, <001>}^* \gg (m_{xz, <001>}^*, m_{xz, <110>}^*, m_{zy, <001>}^*, m_{zy, <110>}^*) \gg m_{xy, <110>}^*$ that, in turn, yields the energy orbital landscape outlined in Fig. 5c,d, which is in agreement with the 2DEG sub-band hierarchy observed in X-ray linear dichroism experiments^{46,49}.

As a consequence of the observed rearrangement of orbital symmetries, the spatial extension of the 2DEG must change significantly with the crystal orientation. In this respect, Fig. 5c,d plot schematically the carrier spatial distributions of d_{xy} , d_{xz} and d_{yz} states: along (001) the first d_{xy} sub-band is expected to be at the bottom of the well, with little spatial spread; on the contrary, along (110) the d_{xy} level raises its energy above the d_{xz}/d_{yz} states, and its spatial extent is considerably larger. In addition to orbital occupancy, contributions from the anisotropic character of the dielectric constant tensor may also influence the 2DEG spatial extent. Therefore, the modulation of the orbital hierarchy described here provides a natural explanation for the distinct anisotropy of the 2D superconductivity and spatial extension for quantum wells oriented along (001) and (110).

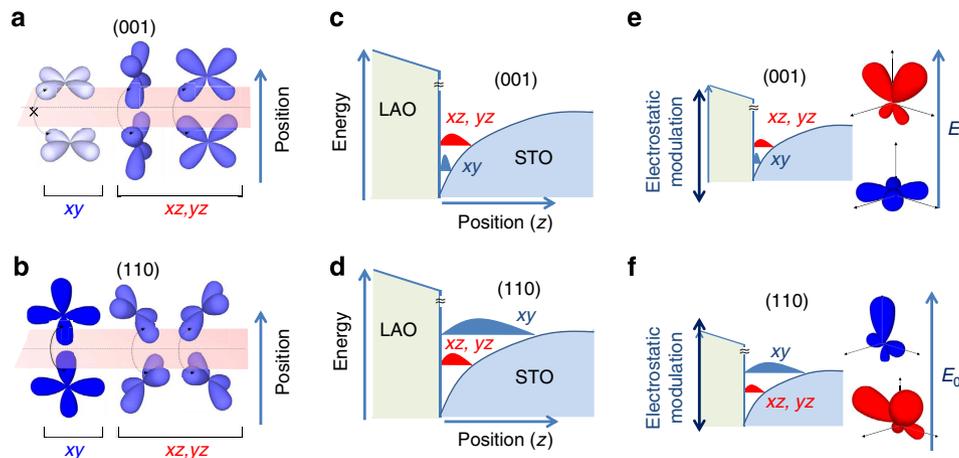


Figure 5 | Energy landscape and orbital symmetries of 2DEGs at (001) and (110) LaAlO₃/SrTiO₃ interfaces. (a) Along (001), the overlapping of d_{xy} orbitals is very small, while it is moderately large for d_{xz}/d_{yz} states. (b) Along (110), the overlapping of d_{xy} orbitals is the largest. Since the effective mass along the confinement direction is inversely proportional to the orbital overlapping, the hierarchy of masses is given by $m_{xy, <001>}^* \gg (m_{xz, <001>}^*, m_{xz, <110>}^*, m_{zy, <001>}^*, m_{zy, <110>}^*) \gg m_{xy, <110>}^*$. The rules of quantum physics in solids dictate the energy landscape in the quantum wells: (c) Orbitals with symmetry d_{xy} lie at the bottom of (001) quantum wells, while d_{xz}/d_{yz} are higher in energy; (d) Along (110), the d_{xz}/d_{yz} states have lower energy and the d_{xy} levels are at the top. The different energy hierarchy of orbitals determines a larger spatial extension for 2DEGs along (110) as compared with (001). (e) Sketches how the d_{xy} and d_{xz}/d_{yz} orbitals are distorted by the inversion symmetry breaking field E_0 at the interface. Along (001), d_{xy} states have a small projection along the out-of-plane direction and they are weakly polarized by E_0 , whereas d_{xz}/d_{yz} states project along the direction of confinement and are largely polarized. (f) Instead, along (110), both kinds of orbitals are affected similarly by the interface fields. Within the range of applied fields, the electrostatic modulation has dissimilar effects on the spin-orbit fields B_{SO} . Along (001), only d_{xy} orbitals are filled at $V_g < 0$, whereas at $V_g > 0$ the d_{xz}/d_{yz} states are progressively occupied. Thus, at low electrostatic doping, B_{SO} is smaller because of the relatively small atomic orbital polarization of d_{xy} states, whereas the population of d_{xz}/d_{yz} orbitals at positive V_g increases significantly B_{SO} . Instead, along (110) the dependence of B_{SO} on the applied fields is weak, because all t_{2g} orbitals are expected to undergo polarizations of similar strength.

By the same token, the redistribution of the orbital sub-band hierarchy also explains the distinctive dependence of the Rashba spin-orbit fields on the orientation. As mentioned above, the interface electric field E_0 induces a polarization of the atomic orbitals that breaks their symmetry along the direction of the quantum well. This enables new covalent channels within the t_{2g} manifold and the oxygen network that contributes to B_{SO} (refs 40,41). The key point is to recognize that orbitals with large projections over the normal to the interface are those more sensitive to the inversion symmetry breaking fields E_0 , giving larger Rashba effects^{40,41}. Such atomic orbital polarization is graphically depicted in Fig. 5e in the form of spatially distorted orbitals.

In the light of these observations, the asymmetric modulation of B_{SO} with field along (001), Fig. 4c, can be explained because d_{xy} orbitals are weakly polarized due to their minimal projection along the confinement direction, while d_{xz}/d_{yz} states have much stronger spatial asymmetry (Fig. 5e). As a result of the 2DEG sub-band hierarchy along (001), the electrostatic modulation of orbital occupancy is anticipated to give a significant variation of B_{SO} as a function of the orbital occupancy: at $V_g < 0$ V only d_{xy} orbitals are populated; the orbital polarization is weak (Fig. 5e) and B_{SO} is relatively small. In contrast, as we enter the regime of accumulation and d_{xz}/d_{yz} bands start to be filled, the spin-orbit term begins to increase significantly, in agreement with the significantly larger orbital polarization of these orbitals.

The situation is radically different for the (110) interface. Now, the electrostatic modulation of B_{SO} is very weak and the spin-orbit field is largely unaffected by the electrostatic gating. This behaviour, which may seem surprising in the light of the modulation of carrier density with electrostatic gating (Fig. 4d), can be well understood on the grounds of the similar atomic orbital polarizations of d_{xy} and d_{xz}/d_{yz} orbitals along (110), Fig. 5e. Indeed, along (110), all t_{2g} orbitals are expected to undergo similarly strong polarizations (Fig. 5e) and, therefore, the spin-orbit field B_{SO} is expected to have a rather weak dependence on the applied field, as confirmed by the experiments.

In summary, we have shown that the orbital reconstruction that occurs for $\text{LaAlO}_3/\text{SrTiO}_3$ quantum wells confined along two different directions, (001) and (110) has a deep impact on the physical properties of these 2DEGs. We claim that the different energy landscapes and hierarchy of orbital symmetries are behind the observed differences in the 2DEG spatial extensions and spin-orbit fields. The analysis of the 2D superconductivity is consistent with 2DEGs extending spatially over (110) at larger distances than at (001) interfaces. At the same time, electrostatic gating experiments have provided relevant clues to understand the distinctive spatial distribution of t_{2g} states with respect to the interface that results from the modified energy sub-band hierarchy and the renormalization of the associated effective band masses. Our work shows that crystal symmetry is an extra degree of freedom to realize different 2DEG band reconstructions at the $\text{LaAlO}_3/\text{SrTiO}_3$ interface, thus allowing a selective occupancy of states of different symmetry. Such new perspective for 2DEG band engineering is very alluring, as it opens new research fields to extend our current understanding of the link between orbital symmetry and magnetism and superconductivity at $\text{LaAlO}_3/\text{SrTiO}_3$ quantum wells.

Methods

Sample preparation. For the growth of (110)-oriented samples, the SrTiO_3 substrates were treated in a dedicated furnace at 1,100 °C for 2 h under ambient conditions^{26,27}. Samples with (001) orientation were grown on TiO_2 -terminated SrTiO_3 substrates. The TiO_2 termination of the $\text{SrTiO}_3(001)$ single crystals was obtained by chemical treatment followed by thermal annealing^{50,51}. LaAlO_3 thin films were grown by pulsed laser deposition ($\lambda = 248$ nm) monitored by high pressure reflection high-energy electron diffraction. The substrates were heated

from room temperature to deposition temperature (850 °C) in an oxygen partial pressure $P_{\text{O}_2} = 0.1$ mbar. During deposition, the LaAlO_3 was grown under a pressure $P_{\text{O}_2} = 10^{-4}$ mbar and 1-Hz repetition rate, with laser pulse energy of around 26 mJ. Films with thickness 7, 8, 10 and 14 MLs were prepared on (110) substrates, whereas the (001)-oriented sample had a LaAlO_3 thickness of 10 MLs. At the end of the deposition, samples were cooled down in an oxygen rich atmosphere to minimize the formation of oxygen vacancies that could lead to extrinsic mechanisms of conduction. More specifically, the samples were cooled from $T = 850$ to 750 °C under a pressure $P_{\text{O}_2} = 0.3$ mbar and under $P_{\text{O}_2} = 200$ mbar from $T = 750$ °C down to room temperature, including a dwell time of 1 h at 600 °C.

Magnetotransport. The electrical characterization was performed by using six-contact arrangement in Hall geometry, from which the sheet resistance, sheet carrier density and electron mobility were extracted as a function of temperature and gate voltage. The current was injected along the in-plane (001) direction in (110)-interfaces. The $\text{LaAlO}_3/\text{SrTiO}_3$ interface was contacted via ultrasonic wire bonder with Al wires. Measurements at temperatures below 1.8 K were measured in a dilution cryostat by applying 50 nA AC current of frequency 13.67 Hz. For the estimation of critical field, the magnetic field was applied parallel and perpendicular to the sample plane with sweep rates of 1.6 mT s^{-1} . For the measurement of the parallel critical magnetic field of the 110 samples, the field was applied in the same direction than the current, that is, along the in-plane (001). Electric fields were applied using voltage source. No leakage current (< 5 nA) was detected up to largest applied voltages ± 400 V.

Transmission electron microscopy. STEM-HAADF images were acquired with a NION UltraSTEM, equipped with a 5th order NION aberration corrector and operated at 200 kV, and in a FEI Titan (60–300 kV) STEM operated at 300 kV, equipped with a probe Cs corrector from CEOS, a monochromator and a high-brightness field-emission gun (X-FEG). HAADF signals for the samples were collected from the detector inner-angles of ~ 86 and ~ 60 mrad for the NION and FEI Titan microscopes, respectively. Specimens for STEM were prepared by conventional methods, by grinding, dimpling and argon ion milling.

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Author contributions

G.H. designed and conceived the experiments with help from J.F., N.B. and J.L. Low-temperature magnetotransport experiments were done by G.S., A.J. and N.B. The analysis of low-temperature magnetotransport data was done by G.S., N.B. and G.H. The thin film preparation, *in situ* reflection high-energy electron diffraction characterization and structural and surface morphology analysis were done by M.S., N.D. and F.S. J.G., and M.V. made all the characterization and analysis of structural characterization by STEM-HAADF. G.H. wrote the paper. All authors discussed the data and commented on the paper.

Additional information

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