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Crystallographically Textured Nanomaterials

Produced from the Liquid Phase Sintering of $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ Nanocrystal Building Blocks

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3 **KEYWORDS:** $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$, nanocrystal, crystal texture, liquid phase sintering, thermoelectric,
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5 energy conversion.
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9 **ABSTRACT:** Bottom-up approaches for producing bulk nanomaterials have traditionally lacked
10 control over the crystallographic alignment of nanograins. This limitation has prevented
11 nanocrystal-based nanomaterials from achieving optimized performances in numerous
12 applications. Here we demonstrate the production of nanostructured $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ alloys with
13 controlled stoichiometry and crystallographic texture through proper selection of the starting
14 building blocks and the adjustment of the nanocrystal-to-nanomaterial consolidation process. In
15 particular, we use disk-shaped $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ nanocrystals and tellurium nanowires as building
16 blocks and hot press the materials using multiple pressure and release steps at a temperature
17 above the tellurium melting point. We explain the formation of the textured nanomaterials
18 through a solution-precipitation mechanism under a uniaxial pressure. Additionally, we further
19 demonstrate these alloys to reach unprecedented thermoelectric figures of merit, up to $ZT=1.96$
20 at 420 K, with an average value of $ZT_{\text{ave}}=1.77$ for the record material in the range 320-500 K,
21 thus potentially allowing up to 60 % higher energy conversion efficiencies than commercial
22 materials.
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45 Nanostructured materials hold the key to the cost-effective use of thermoelectric devices in a
46 broad range of applications. Nanomaterials are characterized by reduced thermal conductivities
47 attributed to phonon scattering at grain boundaries;^{1, 2} the potential to simultaneously combine
48 high charge carrier concentrations with large charge carrier mobilities through modulation
49 doping;³⁻⁵ and the possibility to maximize the Seebeck coefficient by modifying the electronic
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3 band structure or through electron energy filtering at interphases.^{6, 7} However, to optimize all
4 these properties, an exquisite control over nanomaterial parameters in three dimensions is
5 necessary. Such level of control cannot be reached by current high throughput bulk nanomaterial
6 fabrication technologies, such as mechanical grinding. Alternative vacuum-based thin film
7 technologies are able to produce compositionally tuned films, but with limited thickness and at
8 expenses of high production costs and low throughputs.

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18 Bottom-up assembly strategies using nanocrystals as starting building blocks allow producing
19 bulk nanomaterials with parameters tuned at the nanometer-scale and in a cost-effective manner.
20 This approach has been successfully used to produce a plethora of nanocrystalline materials with
21 exceptional thermoelectric properties. However, this methodology faces limitations in the
22 production of highly anisotropic materials with proper crystallographic alignment, as required in
23 numerous application fields, including thermoelectrics.¹

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32 In particular, n-type $\text{Bi}_2\text{Te}_{3-x}\text{Se}_x$ and p-type $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ alloys, the most ubiquitous
33 thermoelectric materials operated at ambient temperature, are layered materials consisting of
34 stacks of covalently bonded quintuple atomic layers, Te-Bi/Sb-Te-Bi/Sb-Te, that are held
35 together by weak van der Waals interactions (Figure S1). Such layered structures are
36 characterized by strongly anisotropic transport properties. In single crystals, the electrical
37 conductivities in the *ab* plane are higher than in the *c* direction, up to factors 4.38 and 2.65 for
38 $\text{Bi}_2\text{Te}_{2.6}\text{Se}_{0.4}$ and $\text{Bi}_{0.5}\text{Sb}_{1.5}\text{Te}_3$ respectively.^{8, 9} Similarly, two-fold higher thermal conductivities
39 are measured within the *ab* plane. Even though the Seebeck coefficient is nearly isotropic,
40 overall, higher thermoelectric figures of merit, up to factors 2.17 and 1.42 for $\text{Bi}_2\text{Te}_{2.6}\text{Se}_{0.4}$ and
41 $\text{Bi}_{0.5}\text{Sb}_{1.5}\text{Te}_3$, are measured in the *ab* plane when compared with the *c* direction. Such strong
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3 anisotropy requires producing crystallographically textured $\text{Bi}_2\text{Te}_{3-x}\text{Se}_x$ and $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ alloys to
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5 further optimize their thermoelectric properties.
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9 To generate crystallographically textured polycrystalline materials, severe plastic deformation
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11 approaches, such as particular extrusion strategies^{10, 11} and high pressure torsion¹² have been
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13 used with some success. More conventional processes, such as hot press and spark plasma
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15 sintering, have been also effective in producing $\text{Bi}_2\text{Te}_{3-x}\text{Se}_x$ and $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ polycrystalline
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17 materials with significant degrees of crystallographic alignment.^{13, 14} Combining these
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19 methodologies with micropowders produced by ball milling, polycrystalline pellets with
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21 relatively thick layered structures, ca. 0.5-1 μm , have been obtained. To optimize such
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23 consolidation processes, most parameters have been investigated, but some controversy has
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25 arisen. As an example, it is generally assumed that anisotropy is stimulated by severe pressures
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27 and temperatures, but in some cases, less orientation at higher processing temperatures have been
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29 reported.¹⁴ In other works the size of the die has been accepted as the main parameter
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31 determining the degree of crystallographic texture.¹² Nevertheless, three main conclusions have
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33 been generally accepted: i) crystallographically aligned $\text{Bi}_2\text{Te}_{3-x}\text{Se}_x$ and $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ alloys are
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35 produced with the c direction oriented along the pressing direction; ii) in such materials,
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37 improved thermoelectric properties are measured or assumed within the ab plane when compared
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39 with the c direction; and iii) much room is left for improvement through optimizing these
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41 processes.
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48 Here, we detail a liquid phase sintering strategy to systematically produce crystallographically
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50 textured $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ materials with thin layered structures. We further prove that these
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52 nanomaterials are characterized by record thermoelectric figures of merit, although not in the ab
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54 plane, but in the c direction.
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3 $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ nanocrystals with controlled composition were used as precursor building blocks to
4 produce bulk nanocrystalline pellets. Figure 1a shows a representative scanning electron
5 microscopy (SEM) micrograph of the disk-shaped $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ nanocrystals produced in
6 diethylene glycol from bismuth (III) nitrate, antimony (III) chloride and sodium tellurite in the
7 presence of hydrazine and polyvinylpyrrolidone (PVP) at 180 °C (see details in the supporting
8 information, SI). Over 5 g of nanocrystals per batch were produced using this synthetic protocol
9 (Figure S2a). Composition could be easily tuned around its optimum value for thermoelectric
10 applications ($x=0.5$) by adjusting the initial ratios of the bismuth and antimony precursors
11 (Figure 1b and table S2-3). In all samples, excess amounts of tellurium, nominally $\text{Te}/(\text{Bi}+\text{Sb})$
12 $=1.875$, were introduced in the reaction mixture. Finally, the produced nanocrystals
13 systematically conserved the nominal Bi/Sb ratios and an excess of tellurium: $\text{Te}/(\text{Bi}+\text{Sb})\sim 1.7$.

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29 $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ nanocrystals were purified by multiple precipitation and redispersion steps (SI).
30 Subsequently, they were dried and annealed at 350 °C for 60 min under argon flow inside a tube
31 furnace to remove remaining organics. The annealed nanopowders still consisted of disk-shaped
32 nanocrystals (Figure 1c and Figure S2b-d). They also contained tellurium nanorods grown during
33 the annealing step from the excess amounts of this element in the original sample, as observed by
34 SEM (Figure 1c) and x-ray diffraction (XRD, Figure 1d).

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44 The annealed nanopowders were hot-pressed for 210 s within an inert atmosphere into
45 cylindrical pellets (10 mm in diameter and 10 mm length) using 80 MPa of pressure. This
46 procedure, lasting less than 5 minutes and requiring no vacuum and moderate processing
47 temperatures, resulted in $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ cylinders with relative densities ca. 90% of the theoretical
48 value, as measured by the Archimedes' method. From these cylinders, rectangular bars of about
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8x6x1 mm³ were cut in two directions, along the press axis, i.e. the cylinder length, and within the plane normal to this axis, i.e. the cylinder disk plane (Figure S4).

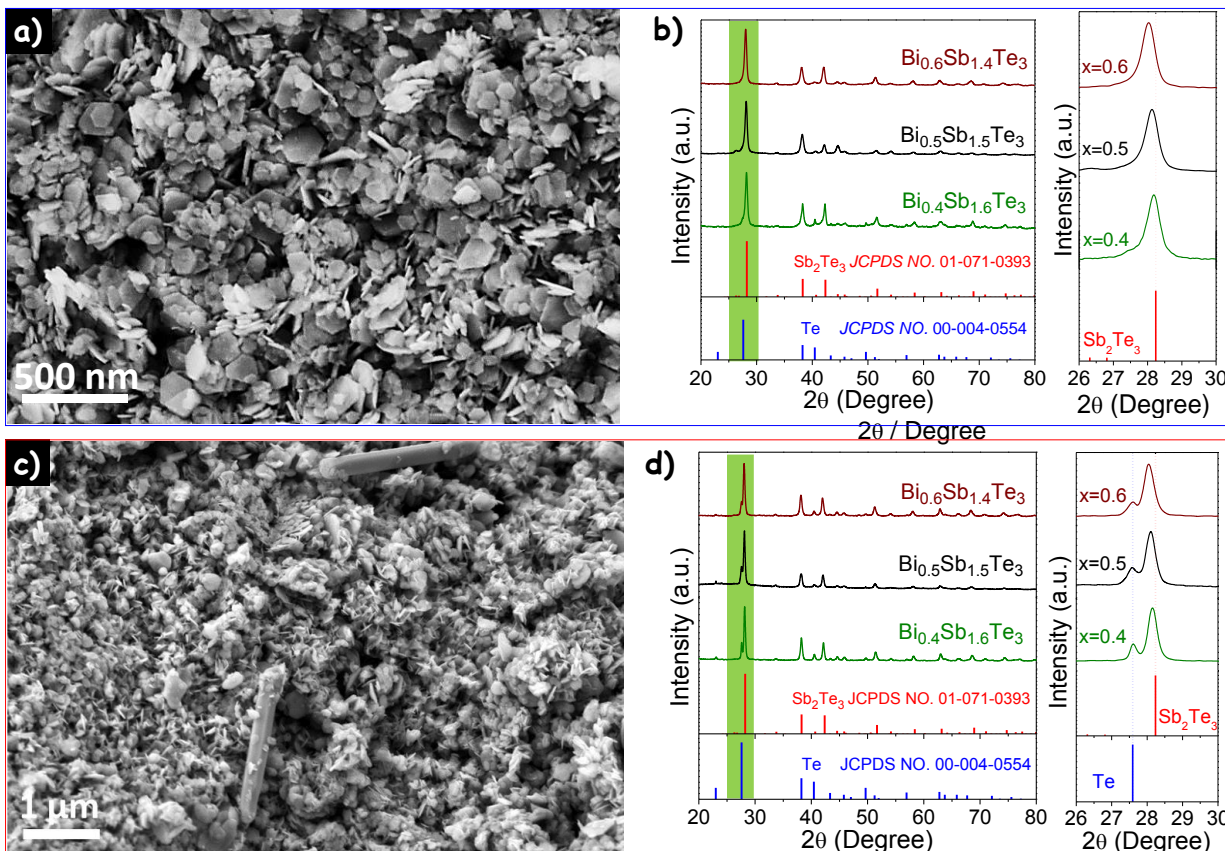


Figure 1. Representative SEM micrographs (a,c) and XRD patterns (b,d) of the Bi_xSb_{2-x}Te₃ disk-shaped nanocrystals used in this work, before (a,b) and after (c,d) annealing. The presence of large tellurium nanorods in the annealed sample is evidenced in the SEM image and from the XRD peak at 2θ=27.6°.

The uniaxial hot press of the materials at temperatures below 400 °C resulted in relatively isotropic pellets containing small nanograins with random orientations (Figure 2f, g). On the other hand, when hot pressing the annealed nanopowders at temperatures above 450 °C, pellets with a strong crystallographic texture, with the [001] crystallographic direction oriented along

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3 the press axis, were produced (Figure 2a-d). Figures 2a-2c show the top-view and cross-section
4 SEM micrographs of a pellet produced from hot pressing $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ nanodisks at 480 °C. A
5 laminar structure, with layer thicknesses of around 40 nm and extending several tens and even
6 hundreds of microns in the disk plane, was systematically observed in all the samples produced
7 at this temperature. Figure 2d and Figure S3 displays the XRD patterns measured in two
8 perpendicular orientations of the cylinder, along and normal to the press axis. The XRD pattern
9 of the pellet held with the press axis normal to the support, showed a clear increase of the
10 diffraction peaks corresponding to the [001] crystallographic directions with respect to the
11 pattern obtained from the cylinder held planar or the precursor nanopowder (Figure 2d).
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24 The excess of tellurium in the initial nanopowder was essential to produce highly
25 crystallographically textured nanomaterials (Figure S5). During the hot press process at 480 °C,
26 the excess of tellurium, ca. 8% in volume, liquefied (tellurium melting point is 449.5 °C),
27 creating a solid-liquid interface with two main potential consequences.
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34 The liquid interface reduced the interparticle friction, potentially facilitating the
35 reaccommodation of the solid nanocrystals. In a pure nanocrystal rearrangement scenario, the
36 final pellet would consist of disk-shaped nanocrystals identical to those observed in the precursor
37 nanopowders, but all aligned in one preferential direction dictated by the press axis. However,
38 this was not the observed pellet nanostructure (Figure 2a-c).
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47 The tellurium liquid phase could also dissolve the randomly oriented $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ nanocrystals,
48 allow a fast atomic diffusion of the alloy elements, and feed, when oversaturated, the nucleation
49 and growth of new $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ crystals oriented in a preferential direction dictated by the press
50 axis. In the liquid phase sintering literature, this sintering mechanism is known as the solution-
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3 reprecipitation stage.¹⁵ Indeed, at 480 °C, tellurium can dissolve in the liquid phase around 15%
4 of Sb and 25% of Bi. Actually, both the Sb-Te and Bi-Te systems have eutectic points: at 422 °C
5 for Sb-Te with a 7.4% of Sb and at 413 °C with 10% of Bi in the Bi-Te system.^{16, 17} The
6 dissolution of the solid in the liquid phase varies inversely with the crystal size, thus smaller
7 grains and higher energy surfaces preferentially dissolve, contributing to the formation of larger
8 grains. In this Ostwald ripening scenario, randomly oriented small disk-shaped nanocrystals are
9 rapidly dissolved and larger and crystallographically oriented structures are produced, strongly
10 reducing the material porosity.^{15, 18} Such nanostructure matched well with the results obtained.

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22 The presence of a uniaxial hydrostatic pressure during the liquid phase sintering decisively
23 influenced the orientation of the growing crystals. During the solution and reprecipitation step,
24 the preferred orientation of the precipitating crystals is the one that minimizes chemical potential
25 required for equilibrium across the plane normal to the pressure axis.¹⁹ Such orientation depends
26 on the elasticity of the material in each crystallographic direction, in such a way that in rather
27 elastically anisotropic materials, such as $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$, crystals align with the weakest
28 crystallographic plane normal to the pressure axis. The layered structure of $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ crystals,
29 with a strong bonding between atoms inside the *ab* plane and weak bonding between the layers
30 in the *c* direction, results in a rather strong anisotropy of the elastic constants.²⁰ In particular, the
31 Young modulus of Bi_2Te_3 crystals in the *a* and *b* axes are close to 30% larger than in the *c* axis.²¹,
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22 Thus, during the liquid phase sintering under a uniaxial pressure, reprecipitated crystals tend to
orientate with the *c* direction parallel to the press axis, as we observed experimentally.

The presence of a uniaxial pressure during the liquid phase sintering may not only introduce a
preferential orientation of the reprecipitated crystals, but it may also influence their geometry.¹⁹

²⁰ In the presence of a liquid phase able to partially dissolve and reprecipitate the crystals, the

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3 uniaxial pressure promotes growth in the directions normal to the uniaxial pressure and aids the
4 dissolution of the crystal in the directions parallel to this axis.^{19, 20} In the particular case of
5 $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$, already in solution, at moderate temperatures and in the absence of a uniaxial
6 pressure, nanocrystals grow faster in the ab plane than in the c -axis, thus acquiring flat, disk-
7 shaped, morphologies. However, during the hot press process, the relative growth in the ab plane
8 is much more accentuated and plates with much higher aspect ratios are finally obtained.
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12 The initial shape of the pressed nanocrystals could influence the direction and even existence of
13 a predominant orientation of the crystal domains in the final material.²³ Actually, in a previous
14 work, no significant crystallographic texture was observed in pellets obtained from the hot press
15 of ball milled $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ containing an excess of tellurium, even when using processing
16 temperatures above the tellurium melting point.²⁴ To determine the role of the geometry of the
17 precursor nanocrystal, we hot-pressed Bi_2Te_3 nanorods with the same excess of tellurium,
18 $\text{Te}/(\text{Bi}+\text{Sb})\sim 1.7$, and following the exact same steps as in the case of $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ nanodisks
19 (Figure S6).²⁵ The polycrystalline material produced had a significant crystallographic texture
20 ($I_{(0015)}/I_{(110)}=5.6$, Figure S7 and S8), but lower than that obtained with the $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ nanodisks
21 ($I_{(0015)}/I_{(110)}=7.3$, Figure 2d). We also produced pellets by hot pressing a powder obtained from
22 the ball milling of a commercial ingot and adding an excess of elemental tellurium to it. These
23 pellets also showed some crystallographic alignment ($I_{(0015)}/I_{(110)}=3.1$, Figures S9 and S10), but
24 also lower than the pellets produced from nanodisks (Figure 2d).
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48 To produce consolidated nanomaterials with stoichiometric composition and having the finest
49 crystallographic texture and optimum thermoelectric properties, it was essential to completely
50 remove the excess amount of tellurium. Conveniently, during the hot press, while the
51 temperature was above the tellurium melting point, the excess of this element was being expelled
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3 through the plunger rod (Figure S11). However, in just one pressure and release step, the
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5 tellurium excess was not completely removed from the sample, probably due to the blocking of
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7 the scape paths within the layered structure or the plunger rod (Figure S12-S14). Only upon
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9 multiple pressure and release steps the excess of tellurium could be totally ejected. Specifically,
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11 we used 5 pressure and release steps during the 210 s that the temperature was set at 480 °C to
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13 completely get rid of the tellurium excess.
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17 Among the different compositions tested, $\text{Bi}_{0.5}\text{Sb}_{1.5}\text{Te}_3$ provided the highest thermoelectric
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19 figures of merit (Figures S15-S18). We associated this experimental fact to a more optimum
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21 charge carrier concentration in this sample. This result is consistent with previous reports and
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23 with the use of this composition in commercial devices. Figure 3 shows the electrical
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25 conductivity, Seebeck coefficient, thermal conductivity and overall thermoelectric figure of merit
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27 of the $\text{Bi}_{0.5}\text{Sb}_{1.5}\text{Te}_3$ nanomaterial hot pressed at 480 °C and measured in two directions, parallel
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29 and normal to the press axis. Displayed results were obtained from averaging experimental data
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31 from five different pellets. These results are compared in figure 3 with those obtained from a
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33 pellet produced with the exact same material, but hot pressed at a lower temperature, 300 °C, and
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35 a commercial ingot measured in the cleavage plane (*ab*) and its normal direction (*c*).
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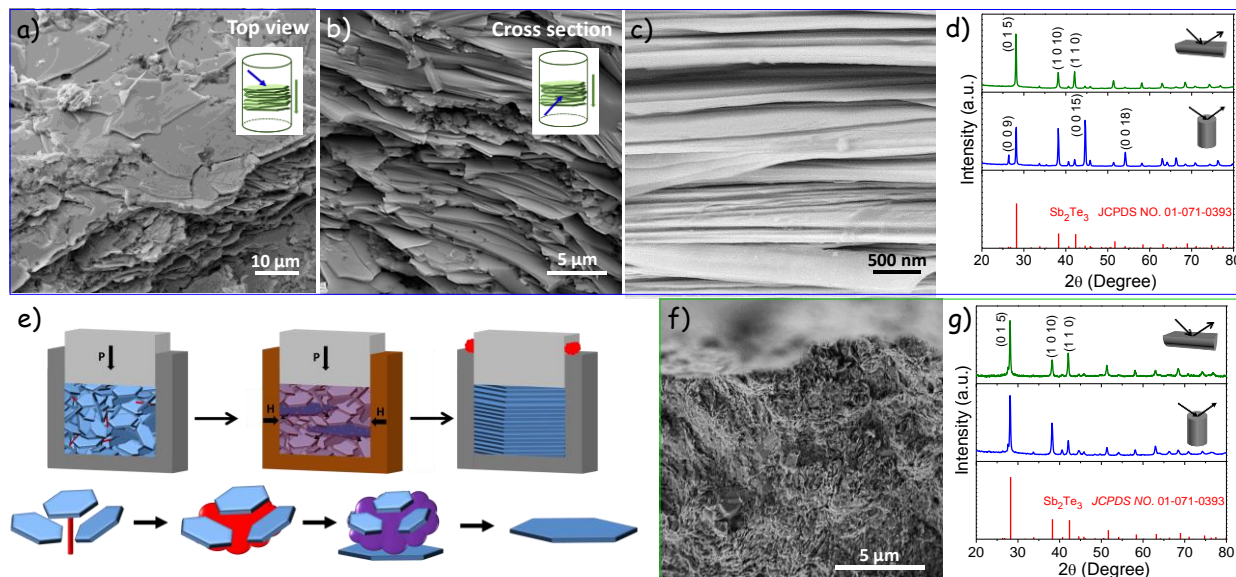


Figure 2. a) Top-view SEM micrograph of a $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ pellet produced at 480 °C. b) Cross-section SEM micrograph of the same pellet. c) Detail of the cross-section SEM micrograph. d) XRD patterns of the pellet placed to have the press axis within the diffraction plane (green) and normal to it (blue). e) Cartoon of the liquid-assisted sintering of the nanocrystals under uniaxial pressure. f) Cross-section SEM micrograph of a pellet hot-pressed at 300 °C. g) XRD patterns of the pellet hot-pressed at 300 °C placed to have the press axis within the diffraction plane (green) and normal to it (blue).

The nanomaterials hot pressed at 300 °C showed relatively low electrical conductivities (Figure 3a). Such low electrical conductivities were associated to a strong scattering of charge carriers at the ubiquitous grain interfaces of these materials. Additionally, these materials had lower relative densities than materials hot pressed at higher temperatures and contained an excess amount of tellurium which had not been expelled during the hot-press step.

On the other hand, nanomaterials hot-pressed at 480 °C showed very high electrical conductivities, comparable to those of the commercial sample. As expected, the electrical

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3 conductivities measured in the disk plane where higher than those obtained in the press direction.
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5 Such high electrical conductivities have been previously measured in samples processed in an
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7 excess of tellurium and have been related to two main factors:^{23, 26} i) an increase (over materials
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9 with similar Bi/Sb ratios) of the hole concentration associated to a modification of the antisite
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11 and vacancy defects due to the tellurium excess;²⁵ ii) formation during the liquid phase sintering
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13 of semicoherent grain boundaries having a minimal effect on hole scattering.²³
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18 Hall measurements of a series of hot-pressed $\text{Bi}_{0.5}\text{Sb}_{1.5}\text{Te}_3$ nanomaterials showed the charge
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20 carrier concentrations to be around $5 \pm 4 \times 10^{19} \text{ cm}^{-3}$, i.e. sensibly higher than that of commercial
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22 materials ca. $1 \times 10^{19} \text{ cm}^{-3}$. The mobilities measured in the plane normal to the press axis were on
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24 average a factor 1.5 higher than those measured in the press axis direction.
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28 The Seebeck coefficients of the bottom-up assembled nanomaterials were significantly lower
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30 than those of the commercial sample, especially for the nanomaterial hot pressed at 480 °C
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32 (Figure 3b). These lower Seebeck coefficients were associated to the higher charge carrier
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34 concentrations present in the bottom-up nanomaterials. The higher charge carrier concentrations
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36 also explained that the Seebeck coefficients started to decrease at a higher temperature in the
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38 nanomaterial hot-pressed at 480 °C when compared with the commercial sample. On top of this
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40 retard of the bipolar effect associated to the higher charge carrier concentration, the excess of
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42 tellurium during the consolidation step could also reduce the density of tellurium vacancies, an
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44 n-type defect, thus additionally decreasing the extent of the bipolar effect and shifting to even
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46 higher temperature the ZT maximum.
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52 Seebeck coefficients measured in each of the two directions were very similar, as expected from
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54 the reduced anisotropy of this parameter already reported for single crystals.⁸ Nevertheless, the
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coefficients measured in the press axis direction were slightly higher, which could point at some extend of selective scattering of the minority carriers at the crystal interfaces in the c direction.

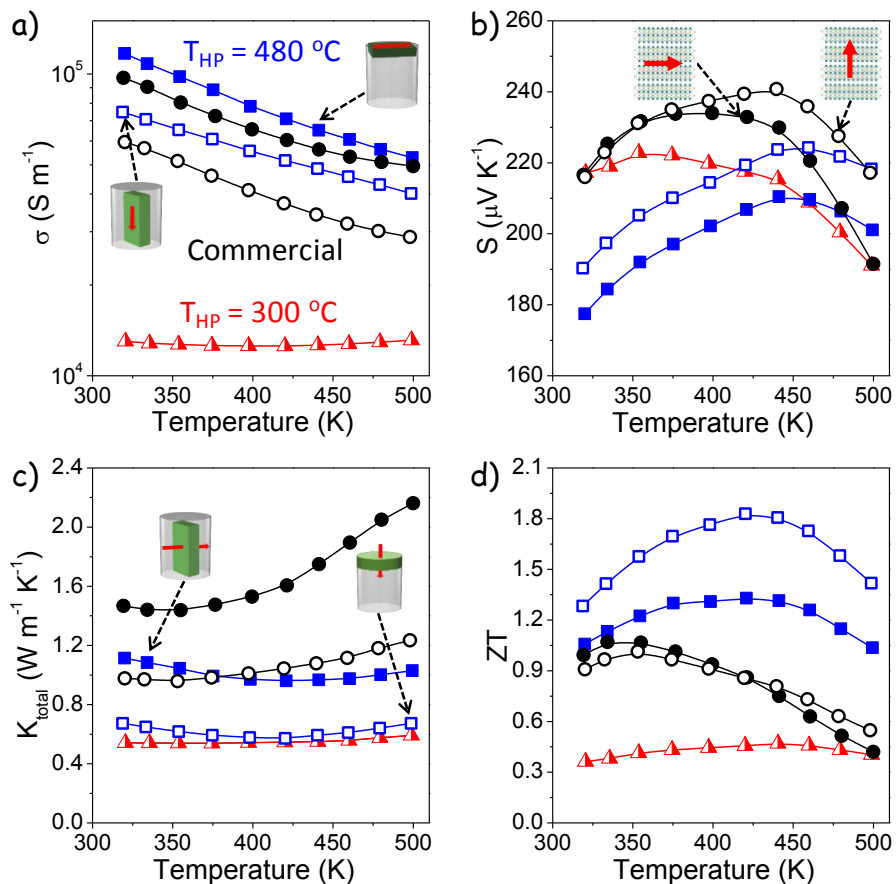


Figure 3. Thermoelectric properties of $\text{Bi}_{0.5}\text{Sb}_{1.5}\text{Te}_3$ nanomaterial hot pressed at 480°C (blue squares) measured in two directions, parallel (open squares) and normal (solid squares) to the press axis (data were obtained from averaging the data values experimentally measured from 5 samples); $\text{Bi}_{0.5}\text{Sb}_{1.5}\text{Te}_3$ nanomaterial hot pressed at 300°C (red triangles); and a commercial ingot (black circles) measured in two directions, parallel (solid circles) and normal (open circles) to the cleavage direction. a) Electrical conductivity, σ ; b) Seebeck coefficient, S ; c) thermal conductivity, κ ; and d) thermoelectric figure of merit, ZT .

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3 As expected, the thermal conductivities of the nanomaterials were significantly lower than those
4 of the commercial sample (Figure 3c). For the nanomaterial hot pressed at 480 °C, the thermal
5 conductivity was particularly low in the direction of the press axis. In this direction, the thermal
6 conductivity of this nanomaterial became comparable to that measured on the pellet hot pressed
7 at 300 °C containing a random distribution of very small crystals. When taking into account the
8 correction for the 10% porosity measured (a factor 1.33), the lowest lattice thermal
9 conductivities that we measured at room temperature were slightly above the amorphous limit
10 for Bi₂Te₃ (0.31 Wm⁻¹ K⁻¹),²⁷ and clearly above those predicted for highly defective Bi₂Te₃ (ca.
11 0.17 Wm⁻¹K⁻¹),²⁸ or for very small Bi₂Te₃ nanograins (0.12 Wm⁻¹K⁻¹)²⁹ and those measured for
12 stacked Bi₂Te₃ 2D films at room temperature (0.1 Wm⁻¹K⁻¹).²⁹
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27 Thermal conductivities decreased with temperature in the low temperature range and stabilized
28 or increased in the high temperature range, when the bipolar contribution became significant
29 (Figures S19). Consistently with previous observations,³⁰ the bipolar contribution to the thermal
30 conductivity was lower in the nanomaterial than in the commercial sample, which was attributed
31 to a combination of the higher majority carrier concentration, the suppression of a source of
32 minority carriers, such as tellurium vacancies, and the scattering of minority carriers at interfaces.
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42 Overall, the thermoelectric figures of merit ($ZT=S^2\sigma T/\kappa$) of the nanomaterial hot pressed at
43 480 °C were significantly higher than those of the commercial sample (Figure 3d). In particular,
44 in the direction of the press axis, the combination of a high electrical conductivity, a notable
45 Seebeck coefficient and a very low thermal conductivity resulted in thermoelectric figures of
46 merit up to $ZT=1.96$ at 420 K for the record material and $ZT=1.83$ when averaged for 5 pellets.
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60 Dispersion of results provided from a dispersion of the charge carrier concentration in the
samples. These values are the highest thermoelectric figure of merit reported in the ambient

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3 temperature range (Figure S20). Importantly, these high thermoelectric figures of merit were
4 sustained in an extended temperature range. The average ZT , calculated over 180 K, from 320 K
5 to 500 K, was $ZT_{\text{ave}}=1.77$ for the record material and $ZT_{\text{ave}}=1.65$ when averaging for 5 pellets.
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7 These values translated in potential energy conversion efficiencies up to 60% higher than those
8 of commercial materials (SI).^{31, 32}
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10 Notice that the figures of merit measured from the nanomaterial in the ab plane were
11 systematically lower than those in the c direction (Figure 3d). This was somehow unexpected,
12 since higher ZT values have been generally measured or assumed in the ab plane for single
13 crystals and partially oriented polycrystalline materials.^{8, 12} The general assumption that the
14 figures of merit in the ab plane must be higher than those in the c direction explain that in several
15 previous works only the thermoelectric properties in the ab plane were measured, when possibly
16 higher figures of merit would have been obtained in the normal direction.^{30, 33}
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19 Finally, we want to highlight that the nanocrystalline $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ materials presented here
20 showed good stability even when heated up to relatively high temperature (≥ 500 °C) and during
21 prolonged periods of time, as can be seen in figures S21-22.
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24 **Conclusions**

25 We demonstrated the production of crystallographically textured $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ bulk nanomaterials
26 with record thermoelectric figures of merit: $ZT=1.96$ for the record sample and $ZT=1.83$ when
27 averaged over 5 materials at 420 K. These high ZT values also extended over a larger
28 temperature range, resulting in ZT values averaged over 180 K as high as: $ZT_{\text{ave}}=1.77$ for the
29 record material and $ZT_{\text{ave}}=1.65$ as the mean value from 5 samples. These unprecedented figures
30 of merit were obtained in the c direction, in contrast with most previous works measuring or
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3 assuming better thermoelectric performances in the *ab* plane. Our procedure was based on
4 several key parameters: i) the use of p-type $\text{Bi}_x\text{Sb}_{2-x}\text{Te}_3$ disk-shaped nanocrystals having
5 controlled composition; b) the uniaxial hot press of the nanocrystals containing an excess of
6 tellurium homogeneously distributed across the material; and c) the use of a proper hot press
7 temperature and processing steps to ensure a proper liquid-assisted sintering and the complete
8 removal of the tellurium excess.
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21 ASSOCIATED CONTENT

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24 **Supporting information** containing details of the chemicals used, NCs synthesis; pellet
25 fabrication, materials characterization (X-ray diffraction patterns, SEM, electrical and thermal
26 characterization); stability measurement and repeatability is available free of charge via the
27 Internet at <http://pubs.acs.org>.
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42 Author Contributions

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46 The manuscript was prepared through the contribution of all authors. D. C. and A. C. guided the
47 project, and supervised the work. Y. L., D. C., M.I., M.K. and A. C. conceived and prepared the
48 manuscript. Y. L., S.O. and D. C. designed the experiments, produced the nanomaterials, and
49 performed the thermoelectric characterization. Y. Z. and K. L. produced the bulk samples. M. I.,
50 K. L. and K. N. performed Hall measurement and analysed the results in this manuscript. A. G.,
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3 S. M., and J. A. performed HRTEM and STEM-EELS and discussed these results. The
4 manuscript was corrected and improved by all authors.
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7 8 **Notes**

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10 The authors declare no competing financial interest.
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TOC graphical abstract

