

Spin States Protected from Intrinsic Electron-Phonon-Coupling Reaching 100 ns Lifetime at Room Temperature in MoSe₂

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We present time-resolved Kerr rotation measurements, showing spin lifetimes of over 100 ns at room temperature in monolayer MoSe₂. These long lifetimes are accompanied by an intriguing temperature dependence of the Kerr amplitude, which increases with temperature up to 50 K and then abruptly switches sign. Using ab initio simulations we explain the latter behavior in terms of the intrinsic electron-phonon coupling and the activation of transitions to secondary valleys. The phonon-assisted scattering of the photo-excited electron-hole pairs prepares a valley spin polarization within the first few ps after laser excitation. The sign of the total valley magnetization, and thus the Kerr amplitude, switches as a function of temperature, as conduction and valence band states exhibit different phonon-mediated inter-valley scattering rates. However, the electron-phonon scattering on the ps time scale does not provide an explanation for the long spin lifetimes. Hence, we deduce that the initial spin polarization must be transferred into spin states which are protected from the intrinsic electron-phonon coupling, and are most likely resident charge carriers which are not part of the itinerant valence or conduction band states.

Monolayers of transition metal dichalcogenides (TMDs) like molybdenum diselenide (MoSe₂) are two-dimensional (2D) semiconductors with a direct band gap and a strong spin orbit splitting of the valence band of several 100 meV. Optical selection rules allow a valley selective excitation of spin states with circular polarized light, which makes this class of material interesting for spintronic applications. $^{2-4}$ The performance of TMDs for spintronics is backed by the measurement of extraordinarily long spin lifetimes in the μ s range both in TMD monolayers and heterostructures at cryogenic temperatures.⁵⁻⁷ Nevertheless, realizing the technological potential of monolayer TMDs will require technologically relevant spin properties at room temperature (RT). In this respect, already the archetype 2D-material, graphene, has demonstrated that it is possible to outperform conventional spintronic materials such as GaAs and Si.⁸ In the case of graphene, spin lifetimes up to 12.6 ns were measured in all-electrical spin precession measurements at RT.⁹ In TMDs, optical Kerr rotation measurements revealed that a spin signal with a lifetime of hundreds of ps can survive up to RT.¹⁰

Here, we present time-resolved Kerr rotation (TRKR) measurements on exfoliated monolayer MoSe₂ flakes, showing room temperature spin lifetimes up to 100 ns. By the combination of energy-dependent, two-color pump-probe Kerr measurements, and photoluminescence spectroscopy on samples with different doping and defect levels, we show that these long lifetimes are likely linked to resident carriers which are not itinerant states. ab initio calculations of electron-phonon coupling show that itinerant carriers are expected to undergo fast phonon-

induced spin relaxation. We show that phonon-induced scattering rates of valence and conduction band states are the key to understand the formation of an initial valley magnetization, which is directly probed by the Kerr rotation. An intriguing increase of the Kerr rotation amplitude, followed by a sign reversal at higher temperatures, can be explained by a temperature-activated change of the conduction and valance band scattering rates with phonons. We argue that the initial laser-prepared magnetization is transferred within ps to long-lived spin states, which are then probed on the ns time scale by Kerr rotation measurements.

The MoSe₂ flakes were mechanically exfoliated from bulk crystals from different suppliers with a polydimethylsiloxane (PDMS) membrane and transferred onto Si/SiO₂ substrates. 11,12 For the TRKR experiments we use two mode-locked Ti:sapphire lasers to independently tune the energies of both pump and probe pulses. An electronic delay between both pulses covers the full laser repetition interval of 12.5 ns. The pulse widths are on the order of 3 ps, the laser spot sizes have a FWHM value of approximately $6-8 \mu m$ and the laser power was kept to $600 \mu W$ for both pump and probe beams. The probe energy was set near the trion peak position, determined for each sample and temperature by photoluminescence measurements, whereas the pump energy was set to a slightly higher energy if not noted otherwise. A detailed scheme of the experimental setup can be found in Ref. 13.

Fig. 1(a) depicts the TRKR traces for the sample showing the unprecedented long spin lifetimes at higher temperatures (sample M1). A spin polarization is created

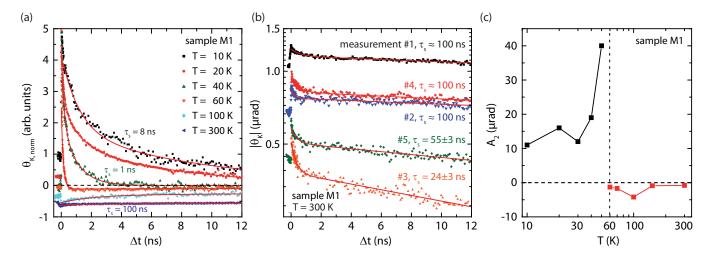


Figure 1. (a) TRKR data of one MoSe₂ monolayer (sample M1) at different temperatures, with bi-exponential fits (red curves), showing spin lifetimes of 100 ns at room temperature (purple curve). While these long-lived spin states appear above 60 K with a Kerr angle of $\Theta_{\rm K} < 0$, there are different spin states with $\Theta_{\rm K} > 0$ at lower temperatures with strongly temperature dependent spin lifetimes. We note that all traces have normalized amplitudes to visualize the overall change in lifetime more clearly. (b) TRKR curves measured at RT on different days yield varying lifetimes from 24 ns to 100 ns. Due to the logarithmic representation of the data, we plot $|\Theta_{\rm K}|$. (c) The amplitude A_2 of the long-lived spin signal from the bi-exponential fit shows an intriguing initial increase with temperature, before it abruptly switches sign at 60 K.

by the circularly polarized pump pulse while the linearly polarized probe pulse measures its temporal decay. The corresponding Kerr rotation $\Theta_K(t)$ is fitted by a biexponential fit function of the form:

$$\Theta_K(t) = \sum_{i=1}^2 A_i \cdot \exp\left(-\frac{t}{\tau_{s,i}}\right),$$
 (1)

where A_i are the amplitudes and $\tau_{s,i}$ are the respective spin lifetimes. This accounts for a rapid initial depolarization which occurs on a time scale of tens to hundreds of ps followed by a slower decay which strongly varies with temperature. If not noted otherwise, all amplitudes and lifetimes in this paper refer to the latter decay.

The overall temporal evolution of Θ_{K} and its temperature dependence up to 40 K are qualitatively similar to many other studies on TMDs, i.e. there are ns-scale spin lifetimes at low temperatures which strongly decrease towards higher temperatures.^{7,10,14–18} It is important to note that even the reported μs spin lifetimes in TMDs can only be measured at cryogenic temperatures and undergo the same strong decay, yielding very short lifetimes at elevated temperatures.⁵⁻⁷ In contrast, we observe strikingly different spin dynamics at higher temperatures: at 40 K (green curve in Fig. 1(a)) $\Theta_{\rm K}$ approaches zero for delay times $\Delta t > 4$ ns, which results from the strong decrease of the spin lifetime. But at higher temperatures an additional spin signal with negative sign emerges. It first appears as a small signal with negative sign for time delays larger than 1 ns at 60 K (orange curve), and becomes fully developed at 100 K at all Δt . Remarkably, there is a very weak temporal decay of Θ_{K} , indicating

extremely long spin lifetimes. Most interestingly, this trend holds over the whole temperature range up to RT, where we observe spin lifetimes of up to 100 ns, 2 orders of magnitude larger than previously reported values at this temperature. We note, however, that the measurements at RT underwent temporal changes on laboratory time scales, which result in a spread of lifetimes ranging from 24 ns up to 100 ns for subsequent measurements taken over several days (see Fig. 1(b), where we plot $|\Theta_{K}|$ on a semi-logarithmic scale for easier comparison). We emphasize that the spin lifetimes did not decrease continuously with each successive measurement, but fluctuated without a discernible pattern (measurements in Fig. 1(b) are labeled in chronological order). Such temporal changes in the spin properties of TMD flakes were already reported previously ^{13,19} and will be discussed further below in more detail.

We note that the amplitudes of the TRKR traces in Fig. 1(a) are normalized in such a way that the discussed overall change in spin lifetimes is most clearly seen. The actual amplitudes are depicted in Fig. 1(c), where we plot the fitted Kerr amplitude A_2 of the longest spin signal seen in the bi-exponential fit at each temperature. Hence, for temperatures below 60 K the black data points represent the spin states with a lifetime which decreases strongly with temperature, whereas for temperatures of 60 K and above, the red data points correspond to the long-lived spin states with negative Kerr amplitudes. The trend of the Kerr amplitude shows a second intriguing feature besides the extraordinary long RT lifetimes: an increase of the Kerr rotation amplitude from 10 K to 50 K, followed by a sign reversal at the temperature at which the long-lived RT spin states emerge.

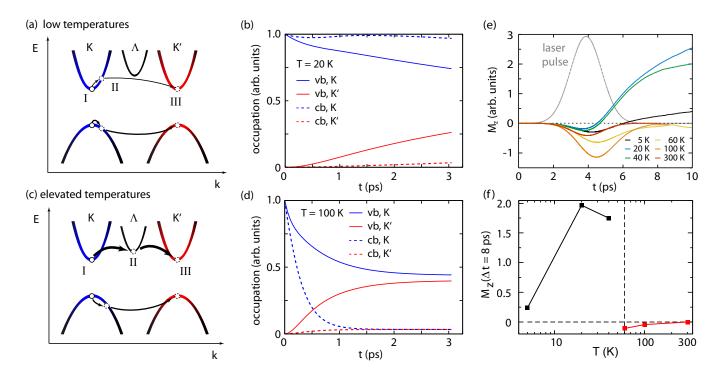


Figure 2. (a) and (c) - Representation of the bands' spin-texture (blue for spin-down, red for spin-up, and black for mixed states) near K and K' at low (a) and elevated (c) temperatures, illustrating the phonon-mediated allowed transitions. White dots mark the electron-hole pair's position in the band structure. At higher temperatures, electrons can transit to an intermediate valley at Λ . (b) and (d) - time-dependent evolution of the integrated occupations of both valence band (vb) and conduction band (cb) in the K and K' valleys at 20 K (b) and 100 K (d) for a prepared ideal exciton. (e) Time evolution of the magnetization at different temperatures when a real pulsed laser field and the e-ph interaction are both turned on. (f) Magnetization at a time delay of $\Delta t = 8$ ps as a function of temperature, showing the initial increase in amplitude and the crossover to a magnetization with reversed sign at around 60 K, matching the experimental data in Fig. 1(c).

Before we explore the spin dynamics of other samples, we first focus on the theoretical understanding of the intriguing temperature dependence of the Kerr rotation amplitude in MoSe₂. The dependence can be understood essentially by changing electron-phonon scattering rates in the conduction and valence bands. We consider intrinsic MoSe₂, and take into account electron-hole pair (exciton) formation by the absorption of a pulsed laser field. We compute the temporal evolution of both conduction and valence band polarizations, solely driven by electron-phonon (e-ph) interactions, by performing real-time calculations with the Yambo code²⁰ (see SI for details). These are based on approximations to the full Baym-Kadanoff equations.^{21–24} In the simulations we use a laser pulse field with parameters (energy, temporal width, and intensity) which track those used in the experiment.

We find a striking resemblance between the temperature-dependence of the total MoSe₂ monolayer magnetization after the pulse and the measured Kerr amplitude. At 60 K (see Fig. 2(e) and 2(f)) there is a complete inversion of the magnetization signal, consistent with the inversion of the Kerr signal shown in Fig. 1(a). This behaviour can be explained by the differences in the dynamics of carriers in the conduction

band at different temperatures (see SI for an extensive analysis). The underlying physical processes are illustrated in Fig. 2 (a). The direct transition between the states at the K and K' points is forbidden by selection rules. However, only the states at band extrema are purely spin-up or spin-down. Those at neighboring wave-vectors contain a mixture of both spins. As a result, inter-valley scattering of a pure spin state requires a two-step electron-phonon process: first, by absorbing an acoustic phonon, carriers are sent away from the band extrema to nearby states (transition from I to II in Fig. 2 (a)); then, temperature permitting, a phonon with a large wave-number can be absorbed or emitted, and the carrier can be transferred to the other valley (transition from II to III in Fig. 2 (a)).

To separate the effects arising from e-ph interaction from those created by the laser field, we first discuss the case of a simple initial state, where we prepare an electron-hole pair exactly at the band extrema in the K valley (further details in the SI). Both carriers are independently allowed to interact with the phonon bath, and we track the time evolution of carrier populations in each band. These are shown in Fig. 2 (b) and (d) where we depict the temporal changes of both conduction (electrons, shown as the dashed lines) and valence band occupations

(holes, solid lines). At low temperatures (Fig. 2 (b)) holes predominantly scatter into the other valley, leading to faster depolarization of the valence bands. The overall magnetization predominantly comes from the electron population imbalance, which barely reduces over the 3 ps time scale.

The situation changes qualitatively at higher temperatures (Fig. 2(d)), where the e-ph scattering rates of the conduction band states have drastically increased. This is largely due to an additional scattering channel through a valley at Λ (also see supplementary Fig. S4) which becomes accessible by thermal excitation (transition from I to II in Fig. 2(c)). This new scattering channel turns out to be faster than those available to the hole population. Electrons will thus depolarize more quickly, leaving behind an imbalanced distribution of holes, which magnetizes the system, but now with a reversed sign. We point out that the valley populations shown in Figs. 2(b) and (d) are obtained by summing all state occupations for a given band near either the K or K' points. As temperature increases, electrons and holes can spread more widely throughout the Brillouin zone, so the total number of electrons and holes inside the region of integration need not to add up to 1.

The final step is the full simulation with both e-ph interaction for different temperatures and the optical excitation by the laser pulse. The resulting time-evolving magnetization is shown in Fig. 2(e). The magnetization is evaluated by summing up over all occupied states in the whole Brillouin zone, taking into account the different spins of electrons and holes. This should not be confused with the valley occupations shown in Figs. 2 (b) and (d). Initially the system has no magnetization, as the laser field creates an equal population of electrons and holes. Only after carriers start to spread through the Brillouin zone, due to e-ph interaction, do we observe an magnetization due to the phonon-induced spin flip processes. At low temperatures (T = 5 K), since holes scatter faster than electrons, the density of electrons is spin polarized $(M_z > 0 \text{ at } 10 \text{ ps})$. Increasing the temperature strongly increases M_z up to 20 K, before it drops and reverses its sign at 60 K ($M_z < 0$ at 10 ps). This behavior nicely matches the Kerr rotation amplitudes in Fig. 1 (c) (see also Fig. 2(f)) and results from the difference in thermal activation of the e-ph scattering rates between electrons and holes, demonstrating that the e-ph interaction is the key to understand the initial formation of the valley polarization in the TRKR measurements.

To support these findings, we next discuss the temperature dependent spin dynamics on five other MoSe₂ samples. The capital letter of the sample's name indicates the supplier ("M" for Manchester Nanomaterials and "H" for HQ Graphene), followed by the sample number indicating the crystal used for fabrication. A final small letter - if present - represents measurements done at different stages of the fabrication process (see below). For all samples we observe the temperature activated increase in valley polarization, which can be seen in the TRKR curves

for samples M2a and H1a in Figs. 3(a) and 3(b), respectively. At first glance, there is a continuous decrease of the Kerr rotation amplitude with increasing temperature. In this context, however, it is important to note that, in contrast to sample M1 (Fig. 1(a)), the other samples have spin lifetimes of more than 100 ns at 10 K (Fig. 3(c)), which is much longer than the laser repetition interval of 12.5 ns. Therefore, the Kerr rotation signal from previous pump pulses is not fully decayed prior to the arrival of the pump pulse at $\Delta t = 0$ ns (see non-vanishing Θ_K for time delays $\Delta t < 0$ ns in Figs. 3(a) and 3(b)). As a result, the constructive superposition of the valley polarization from successive pulses increases the overall Kerr amplitude. As the spin lifetime decreases with temperature (see Fig. 3(c)), the residual valley polarization from previous pump pulses also drops significantly, which explains the overall decrease in the Kerr amplitude with temperature as seen in Figs. 3(a) and 3(b).

We now focus on the additional valley polarization created by the pump pulses at $\Delta t = 0$ ns, which is highlighted by arrows in Figs. 3(a) and 3(b). The gain of the Kerr rotation amplitude first increases with temperature, then it strongly decreases at around 60 K (see light blue curves) for all samples. This temperature corresponds very well to the crossover predicted by our simulation in Fig. 2(f) and to the data of sample M1 in Fig. 1(c), demonstrating that the temperature-dependent increase of the valley polarization probed by the Kerr effect is a hallmark for the spin dynamics in MoSe₂. We note, however, that we do not observe the sign reversal of the Kerr signal at higher temperatures, or the long lived RT spin signal, in any other sample. In contrast, the spin lifetime in those samples dramatically decreases at high temperatures, reaching values on the order of 100 ps at 100 K (Fig. 3(c)).

To unravel why only sample M1 shows the intriguing high temperature spin dynamics, we conduct photoluminescence (PL) measurements on all samples, as summarized in Fig. 4. For PL excitation we use a continuous wave (cw)-laser with an energy of 2.33 eV, at a similar laser power and spot size as used for TRKR measurements. The spectra in Figs. 4(a) to 4(c) show typical emission from neutral (X_0) and from charged excitons $(X_{+/-})$ which were fitted by Voigt functions (dotted lines in Fig. 4(a)) to extract both their width (Figs. 4(d)) and energy position (Figs. 4(e)).

Details in the sample fabrication process have a strong impact on the excitonic properties of the samples, as seen for samples H1a and M2a in Figs. 4(b) and 4(c). These two samples were fabricated from two different crystals from different suppliers, but the samples were fabricated simultaneously under identical conditions, i.e. using the same exfoliation method, the same chemicals for the same duration of time, and were stored and measured identically. Both samples show low-temperature spin lifetimes of over 100 ns, which are, to our knowledge, the longest lifetimes reported so far for $MoSe_2$ at these temperatures (see Fig. 3(c)). Interestingly, the absolute values of the

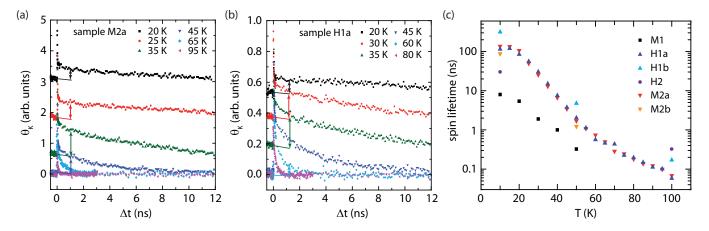


Figure 3. (a) and (b): Temperature dependent TRKR traces for MoSe₂ samples M2a and H1a, respectively. The overall Kerr amplitude decreases as there is less remaining valley polarization from previous pump pulses (see Θ_K signal for time delays $\Delta t < 0$ ns) at elevated temperatures due to the decrease in spin lifetime. In contrast, the Kerr amplitude induced by the pump pulse at $\Delta t = 0$ ns (see arrows) first increases with increasing temperature before it drops significantly at higher temperatures. (c) Spin lifetime of the long-lived spin signal from the bi-exponential fit ($\tau_{s,2}$) of all MoSe₂ samples vs. temperature. For sample M1 we only show low temperature spin lifetimes, which interestingly are the shortest of all samples.

spin lifetimes and their temperature dependence are almost identical (Fig. 3(c), blue and red triangles) although their PL spectra differ qualitatively: the intensity ratio between neutral and charged exciton emission inverts, and a low energy tail in the spectra of sample H1a is observed.

The impact of further post processing steps on the samples' properties is also seen in Figs. 4(b) and 4(c). For sample H1a, we used a hot pick up process, which is known for its surface cleaning effect, 25 to deposit a thin hBN flake on top of the MoSe₂ monolayer. As a result, sample H1b shows a significant reduction in the PL line widths and the vanishing of the low-energy tail (Fig. 4(b)). Independent of temperature, this encapsulation yields an overall increase of the spin lifetimes by a factor of three (Fig. 3(c), dark blue and light blue triangles). In contrast, sample M2b, which was made from sample M2a by annealing under ambient conditions at 180°C, shows additional defects as seen by the appearance of a low-energy tail (Fig. 4(c)). Simultaneously, we observe a decrease in measured spin lifetime after annealing (compare red and orange triangles in Fig. 3(c)). Overall, these results show that, as could be expected, samples with less disorder (e.g. defects or adsorbates) show longer spin lifetimes in TRKR measurements at low temperatures. We have seen that both the spin lifetimes and the excitonic features strongly depend on details in the fabrication process, indicating that the electronic properties are dominated by extrinsic effects. This is a probable explanation for the significant device-to-device variations found in literature.

Sample M1 also follows the trend between disorder seen in PL and measured spin lifetime in TRKR: On the one hand, the low temperature spin lifetime is the lowest measured for all our samples (Fig. 3(c)), on the other hand, this sample shows by far the largest PL peak widths

(Fig. 4(d)). These large peak widths are an indication of a significant variability in the potential landscape of the flake, induced either by disorder in the crystal lattice or by impurities. 26-29 However, there is also a striking difference in sample M1, which becomes obvious when comparing Figs. 4(d) and 4(e). For the other samples a narrower peak width is accompanied by a larger blue shift of the exciton peak position (see dashed lines, which are guides to the eyes). Despite the significant peak width of sample M1, its exciton peak position is the most blueshifted among all samples, by 1.665 eV which is higher than the vast majority of values reported in literature.³⁰ It was shown in several studies that either doping, 31-33strain, $^{34-36}$ or the change of the dielectric environment by encapsulation with hBN^{27,30,37} can be associated with a red-shift of the exciton peak. As no sample except for H1b is in contact with hBN, and all samples are put on Si/SiO₂ substrates, we exclude the dielectric environment as a reason for the large blue-shift: the high exciton energy of sample M1 is therefore indicative of low overall doping and strain.

The large PL line widths of sample M1 indicate the presence of defects, but these do not carrier-dope the sample, which is apparent in the blue-shifted exciton peak and also the fact that the neutral exciton is much more pronounced than the trion in Fig. 4(a). The impact of defects on the electronic structure of sample M1 can also be seen in an unconventional temperature dependent quantum yield, shown in Fig. 4(f) where we normalized the total counts of the exciton peak by the integration time of the spectrometer and the power of the excitation laser. For comparison, sample H2 shows the expected temperature-dependent decrease in quantum yield for MoSe₂, which is attributed to the thermal activation of dark excitons.^{29,38} On the other hand, the quantum yield of sample M1 is orders of magnitude lower at cryo-

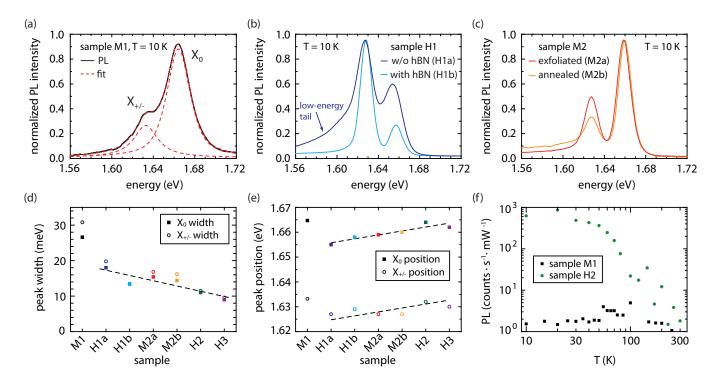


Figure 4. (a)-(c): Photoluminescence spectra of different samples measured at 10 K. The spectra were fitted with Voigtfunctions (dashed lines in (a)) to determine the peak positions and widths of both the neutral exciton (X_0) and the charged trion $(X_{+/-})$. The corresponding fit results for all of our samples at 10 K are shown in (d) and (e), respectively. Here, it can be seen that the sample with the long-lived room temperature spin signal (sample M1) has by far the largest peak width, but at the same time exhibits the most blue-shifted peak positions (dashed lines are guides to the eyes). Sample M1 also distinguishes itself from other samples by showing a pronounced suppression of quantum yield over the whole temperature range, as seen in (f) where the normalized PL intensity of the neutral exciton peak is depicted for samples M1 and H2.

genic temperatures and shows only a negligible temperature dependence. This behavior can be explained by defects which provide non-radiative recombination channels for the photo-excited charge carriers. 29

To explore the origin of the extraordinary hightemperature spin dynamics in sample M1, we conducted energy-dependent Kerr rotation measurements, where the pump energy was scanned across the exciton energy range at different temperatures (see Fig. 5). The Kerr amplitude of the nanosecond spin signal (symbols) follows the PL spectra (solid lines) only at low temperatures. This resonant behavior weakens with increasing temperatures, and finally vanishes for $T > 50 \,\mathrm{K}$. At this point, the Kerr rotation amplitudes correspond to the high temperature long-lived spin signal with inverted sign (red data points in Fig. 1(c)). At temperatures below 50 K the spin dynamics in sample M1 seems to be directly connected to the excitons. The fact that the pump scans at higher temperatures are almost energy independent indicates that the spin states are no longer hosted by the excitons. Rather, this indicates that there has been a transfer of spin polarization from excitons into other long-lived states.

By combining all our findings, a complete picture of the spin dynamics observed in sample M1 can be obtained. The high-energy exciton peak position (Fig. 4(e))

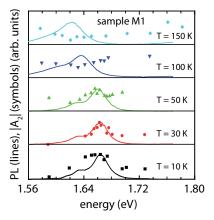


Figure 5. Comparison between PL spectra and amplitude $|A_2|$ of the long-lived Kerr rotation signal at varying pump energies for different temperatures. Whereas the pump scans follow the PL spectra for low temperatures, the excitonic energy resonance in the Kerr measurements vanishes for higher temperatures. Each horizontal line is the zero baseline for both the pump scan and the PL spectrum right above it.

indicates an overall small combination of strain and doping. This justifies our simulations of the temperaturedependent spin dynamics, where we assumed an intrinsic, undoped monolayer of MoSe₂ (Fig. 2). On the one hand, our simulations can explain the intriguing temperature-dependent Kerr amplitude of sample M1 seen in Fig. 1(c), which is due to the change of the ratio between phonon-induced conduction and valance band scattering rates. But on the other hand, our simulations predict that the magnetization drops to zero within a few ps at 300 K (Fig. 2(e)). At lower temperatures we expect that the strong e-ph coupling in TMDs³⁹⁻⁴³ will lead to a complete depolarization of the magnetization of itinerant states, both in the conduction and valence band, somewhere in the ps range. This was also the order of magnitude found in WSe₂.³⁹ All these time scales are much shorter than the measured ns spin lifetimes, whether at high or low T.

The solution to this apparent discrepancy is that the strong electron-phonon scattering mechanism depicted in Fig. 2 only serves to prepare an initial spin state on the ps time scale. Accordingly, part of the initial magnetization must scatter into spin states which are protected over a ns time scale from the strong electron-phonon interaction of itinerant band states. In this respect it is important that excitons can be localized by trapping. 44,45 The capture time by defects is measured to be around 1 ps, 46 which is well within the lifetime of our simulated magnetization process. Furthermore, defect-bound excitons in monolayer TMDs are found to exhibit photoluminescence lifetimes in the hundreds of ns range. 45,47 However. this picture only holds for low temperatures: for temperatures approaching 100 K the thermal energy exceeds the typical localization energy of excitons in such traps. 44,48 This explains the strong decrease of spin lifetimes with temperature up to 100 K.

Another possible explanation for the ns lifetimes at low temperatures might be connected to polarons, which recently have drawn quite some attention in TMDs. ^{32,40,49–52} We think that it is reasonable to assume that (dark) excitons ⁵³ can be protected against the strong electron-phonon interaction through a polaronic reconstruction of the combined electron and phonon system, which goes beyond the theory implemented in our real time simulations. Both explanations (trapped excitons and polaron-exciton complexes) depend on excitons and therefore would explain why the Kerr amplitude at low temperatures follows the excitonic features in the PL spectra of Fig. 5.

The long-lived spin states at higher temperatures, however, do not scale with the number of excitons created, as seen in Fig. 5, and therefore must have another origin. We first note that due to different inter-valley scattering rates of electrons and holes, their polarization can be transferred to resident charge carriers as soon as the excitons recombine. As explained above, we do not expect that these resident charge carriers exhibiting ns spin lifetimes are itinerant states of either the conduction or the valence band. Instead, a recent combined scanning tunneling spectroscopy and photoluminescence study identified defect states in the band gap of monolayer MoSe₂

samples which show a suppressed quantum yield similar to our sample M1 in Fig. 4(f).²⁹ The large peak widths of the PL spectra in sample M1 are also an indication of large spatial fluctuations of the electrostatic potential, which could create puddles of hole or electron doping. Because of the overall small doping of sample M1, these puddles may be well separated. We imagine these puddles to be similar to those in graphene, 54 and in fact first signatures of localized charge puddles have been seen in μ PL measurements of TMD materials.⁵⁵ Defect states in these charge puddles would result in strong localization, which may protect the spin from phonon-induced scattering. Such a picture can also explain the strong variations in the repeated room temperature measurements in Fig. 1(b), as the constant desorption and absorption of volatile gas species at RT constantly changes the charge puddles.

In conclusion, we have demonstrated that electronphonon interaction is the key for the ultrafast formation of a valley polarization in MoSe₂ after optical excitation by circularly polarized light. Different phononinduced scattering rates of both valence and conduction band states were shown to result in a temperature dependent reversal of the valley polarization in good agreement to measurements from time-resolved Kerr rotation. While electron-phonon scattering involving itinerant band states is expected to result in a rapid depolarization, the measured spin lifetimes of up to 100 ns both at low temperatures and at room temperature indicate that the spin polarization scatters into spin states which are protected from intrinsic electron-phonon interaction.

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- ¹ Z. Y. Zhu, Y. C. Cheng, and U. Schwingenschlögl, Phys. Rev. B **84**, 153402 (2011).
- ² D. Xiao, G.-B. Liu, W. Feng, X. Xu, and W. Yao, Phys. Rev. Lett. **108**, 196802 (2012).
- ³ X. Xu, W. Yao, D. Xiao, and T. F. Heinz, Nat. Phys. **10**, 343 (2014).
- ⁴ M. Gmitra and J. Fabian, Phys. Rev. B **92**, 155403 (2015).
- ⁵ P. Dey, L. Yang, C. Robert, G. Wang, B. Urbaszek, X. Marie, and S. A. Crooker, Phys. Rev. Lett. **119**, 137401 (2017).
- ⁶ C. Jin, J. Kim, M. I. B. Utama, E. C. Regan, H. Kleemann, H. Cai, Y. Shen, M. J. Shinner, A. Sengupta, K. Watanabe, T. Taniguchi, S. Tongay, A. Zettl, and F. Wang, Science 360, 893 (2018).
- ⁷ J. Kim, C. Jin, B. Chen, H. Cai, T. Zhao, P. Lee, S. Kahn, K. Watanabe, T. Taniguchi, S. Tongay, M. F. Crommie, and F. Wang, Sci. Adv. 3, e1700518 (2017).
- S. Roche, J. Åkerman, B. Beschoten, J.-C. Charlier, M. Chshiev, S. P. Dash, B. Dlubak, J. Fabian, A. Fert, M. Guimaraes, F. Guinea, I. Grigorieva, C. Schönenberger, P. Seneor, C. Stampfer, S. O. Valenzuela, X. Waintal, and B. van Wees, 2D Mater. 2, 030202 (2015).
- ⁹ M. Drögeler, C. Franzen, F. Volmer, T. Pohlmann, L. Banszerus, M. Wolter, K. Watanabe, T. Taniguchi, C. Stampfer, and B. Beschoten, Nano Lett. **16**, 3533 (2016).
- W.-T. Hsu, Y.-L. Chen, C.-H. Chen, P.-S. Liu, T.-H. Hou, L.-J. Li, and W.-H. Chang, Nat. Commun. 6, 8963 (2015).
- ¹¹ A. Castellanos-Gomez, M. Buscema, R. Molenaar, V. Singh, L. Janssen, H. S. J. van der Zant, and G. A. Steele, 2D Mater. 1, 011002 (2014).
- ¹² L. Banszerus, H. Janssen, M. Otto, A. Epping, T. Taniguchi, K. Watanabe, B. Beschoten, D. Neumaier, and C. Stampfer, 2D Mater. 4, 025030 (2017).
- ¹³ F. Volmer, S. Pissinger, M. Ersfeld, S. Kuhlen, C. Stampfer, and B. Beschoten, Phys. Rev. B 95, 235408 (2017).
- ¹⁴ C. R. Zhu, K. Zhang, M. Glazov, B. Urbaszek, T. Amand, Z. W. Ji, B. L. Liu, and X. Marie, Phys. Rev. B **90**, 161302 (2014).
- ¹⁵ X. Song, S. Xie, K. Kang, J. Park, and V. Sih, Nano Lett. 16, 5010 (2016).
- ¹⁶ L. Yang, N. A. Sinitsyn, W. Chen, J. Yuan, J. Zhang, J. Lou, and S. A. Crooker, Nat. Phys. **11**, 830 (2015).
- ¹⁷ L. Yang, W. Chen, K. M. McCreary, B. T. Jonker, J. Lou, and S. A. Crooker, Nano Lett. **15**, 8250 (2015).
- ¹⁸ M. Goryca, N. P. Wilson, P. Dey, X. Xu, and S. A. Crooker, Science Advances 5, eaau4899 (2019).
- ¹⁹ M. Schwemmer, P. Nagler, A. Hanninger, C. Schüller, and T. Korn, Appl. Phys. Lett. 111, 082404 (2017).
- D. Sangalli, A. Ferretti, H. Miranda, C. Attaccalite, I. Marri, E. Cannuccia, P. M. Melo, M. Marsili, F. Paleari, A. Marrazzo, G. Prandini, P. Bonfá, M. O. Atambo, F. Affinito, M. Palummo, A. M. Sanchez, C. Hogan, M. Grüning, D. Varsano, and A. Marini, Journal of Physics: Condensed Matter (2019).
- ²¹ P. M. M. C. de Melo and A. Marini, Phys. Rev. B 93, 155102 (2016).
- ²² A. Marini, J. Phys. Conf. Ser. **427**, 012003 (2013).
- ²³ C. Attaccalite, M. Grüning, and A. Marini, Phys. Rev. B 84, 245110 (2011).

- ²⁴ E. Perfetto, D. Sangalli, A. Marini, and G. Stefanucci, Phys. Rev. B **92**, 205304 (2015).
- D. G. Purdie, N. M. Pugno, T. Taniguchi, K. Watanabe, A. C. Ferrari, and A. Lombardo, Nat. Commun. 9, 5387 (2018).
- ²⁶ G. Moody, C. Kavir Dass, K. Hao, C.-H. Chen, L.-J. Li, A. Singh, K. Tran, G. Clark, X. Xu, G. Berghäuser, E. Malic, A. Knorr, and X. Li, Nat. Commun. 6, 8315 (2015).
- ²⁷ O. A. Ajayi, J. V. Ardelean, G. D. Shepard, J. Wang, A. Antony, T. Taniguchi, K. Watanabe, T. F. Heinz, S. Strauf, X.-Y. Zhu, and J. C. Hone, 2D Mater. 4, 031011 (2017).
- F. Cadiz, E. Courtade, C. Robert, G. Wang, Y. Shen, H. Cai, T. Taniguchi, K. Watanabe, H. Carrere, D. Lagarde, M. Manca, T. Amand, P. Renucci, S. Tongay, X. Marie, and B. Urbaszek, Phys. Rev. X 7, 021026 (2017).
- D. Edelberg, D. Rhodes, A. Kerelsky, B. Kim, J. Wang, A. Zangiabadi, C. Kim, A. Abhinandan, J. Ardelean, M. Scully, D. Scullion, L. Embon, I. Zhang, R. Zu, E. J. G. Santos, L. Balicas, C. Marianetti, K. Barmak, X.-Y. Zhu, J. Hone, and A. N. Pasupathy, arXiv (2018), 1805.00127.
- ³⁰ See Supplemental Material for additional PL and Kerr data on other samples, magnetic field dependent measurements, information about the fitting procedure, and an in-depth discussion of our theoretical model.
- ³¹ Y. Zhou, G. Scuri, D. S. Wild, A. A. High, A. Dibos, L. A. Jauregui, C. Shu, K. De Greve, K. Pistunova, A. Y. Joe, T. Taniguchi, K. Watanabe, P. Kim, M. D. Lukin, and H. Park, Nat. Nanotechnol. 12, 856 (2017).
- ³² M. Sidler, P. Back, O. Cotlet, A. Srivastava, T. Fink, M. Kroner, E. Demler, and A. Imamoglu, Nat. Phys. 13, 255 (2016).
- ³³ A. M. Jones, H. Yu, N. J. Ghimire, S. Wu, G. Aivazian, J. S. Ross, B. Zhao, J. Yan, D. G. Mandrus, D. Xiao, W. Yao, and X. Xu, Nat. Nanotechnol. 8, 634 (2013).
- ³⁴ J. O. Island, A. Kuc, E. H. Diependaal, R. Bratschitsch, H. S. J. van der Zant, T. Heine, and A. Castellanos-Gomez, Nanoscale 8, 2589 (2016).
- ³⁵ C. R. Zhu, G. Wang, B. L. Liu, X. Marie, X. F. Qiao, X. Zhang, X. X. Wu, H. Fan, P. H. Tan, T. Amand, and B. Urbaszek, Phys. Rev. B 88, 121301 (2013).
- ³⁶ H. J. Conley, B. Wang, J. I. Ziegler, R. F. Haglund, S. T. Pantelides, and K. I. Bolotin, Nano Lett. 13, 3626 (2013).
- ³⁷ J. Wierzbowski, J. Klein, F. Sigger, C. Straubinger, M. Kremser, T. Taniguchi, K. Watanabe, U. Wurstbauer, A. W. Holleitner, M. Kaniber, K. Müller, and J. J. Finley, Sci. Rep. 7, 12383 (2017).
- ³⁸ G. Wang, C. Robert, A. Suslu, B. Chen, S. Yang, S. Alamdari, I. C. Gerber, T. Amand, X. Marie, S. Tongay, and B. Urbaszek, Nat. Commun. 6, 10110 (2015).
- ³⁹ A. Molina-Sánchez, D. Sangalli, L. Wirtz, and A. Marini, Nano Lett. 17, 4549 (2017).
- ⁴⁰ D. Christiansen, M. Selig, G. Berghäuser, R. Schmidt, I. Niehues, R. Schneider, A. Arora, S. M. de Vasconcellos, R. Bratschitsch, E. Malic, and A. Knorr, Phys. Rev. Lett. **119**, 187402 (2017).
- ⁴¹ B. R. Carvalho, Y. Wang, S. Mignuzzi, D. Roy, M. Terrones, C. Fantini, V. H. Crespi, L. M. Malard, and M. A. Pimenta, Nat. Commun. 8, 14670 (2017).
- ⁴² C. M. Chow, H. Yu, A. M. Jones, J. R. Schaibley,

M. Koehler, D. G. Mandrus, R. Merlin, W. Yao, and X. Xu, npj 2D Mater. Appl. 1, 33 (2017).

⁴³ S. Shree, M. Semina, C. Robert, B. Han, T. Amand, A. Balocchi, M. Manca, E. Courtade, X. Marie, T. Taniguchi, K. Watanabe, M. M. Glazov, and B. Urbaszek, Phys. Rev. B 98, 035302 (2018).

⁴⁴ T. Godde, D. Schmidt, J. Schmutzler, M. Aßmann, J. Debus, F. Withers, E. M. Alexeev, O. Del Pozo-Zamudio, O. V. Skrypka, K. S. Novoselov, M. Bayer, and A. I. Tartakovskii, Phys. Rev. B **94**, 165301 (2016).

⁴⁵ A. J. Goodman, A. P. Willard, and W. A. Tisdale, Phys. Rev. B **96**, 121404 (2017).

⁴⁶ K. Chen, R. Ghosh, X. Meng, A. Roy, J.-S. Kim, F. He, S. C. Mason, X. Xu, J.-F. Lin, D. Akinwande, S. K. Banerjee, and Y. Wang, npj 2D Mater. Appl. 1, 15 (2017).

⁴⁷ G. Moody, K. Tran, X. Lu, T. Autry, J. M. Fraser, R. P. Mirin, L. Yang, X. Li, and K. L. Silverman, Phys. Rev. Lett. **121**, 057403 (2018).

⁴⁸ T. Yan, X. Qiao, X. Liu, P. Tan, and X. Zhang, Appl. Phys. Lett. **105**, 101901 (2014).

- ⁴⁹ F. Caruso, H. Lambert, and F. Giustino, Phys. Rev. Lett. 114, 146404 (2015).
- ⁵⁰ D. K. Efimkin and A. H. MacDonald, Phys. Rev. B 95, 035417 (2017).
- ⁵¹ P.-F. Li and Z.-W. Wang, J. Appl. Phys. **123**, 204308 (2018).
- ⁵² Q. Chen, W. Wang, and F. M. Peeters, J. Appl. Phys. 123, 214303 (2018).
- ⁵³ E. Malic, M. Selig, M. Feierabend, S. Brem, D. Christiansen, F. Wendler, A. Knorr, and G. Berghäuser, Physical Review Materials 2, 014002 (2018).
- J. Martin, N. Akerman, G. Ulbricht, T. Lohmann, J. H. Smet, K. von Klitzing, and A. Yacoby, Nat. Phys. 4, 144 (2008).
- N. J. Borys, E. S. Barnard, S. Gao, K. Yao, W. Bao, A. Buyanin, Y. Zhang, S. Tongay, C. Ko, J. Suh, A. Weber-Bargioni, J. Wu, L. Yang, and P. J. Schuck, ACS Nano 11, 2115 (2017).
- ⁵⁶ Research Center Jülich GmbH, HNF Helmholtz Nano Facility, J. Large Scale Res. Facil. (JLSRF) 3, A112 (2017).