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Thermal conductivity and phonon hydrodynamics in transition metal dichalcogenides from first-principles

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Abstract. We carry out a systematic study of the thermal conductivity of four single-layer transition metal dichalcogenides, MX_2 ($M = Mo, W$; $X = S, Se$) from first-principles by solving the Boltzmann Transport Equation (BTE). We compare three different theoretical frameworks to solve the BTE beyond the Relaxation Time Approximation (RTA), using the same set of interatomic force constants computed within density functional theory (DFT), finding that the RTA severely underpredicts the thermal conductivity of MS_2 materials. Calculations of the different phonon scattering relaxation times of the main collision mechanisms and their corresponding mean free paths (MFP) allow evaluating the expected hydrodynamic behaviour in the heat transport of such monolayers. These calculations indicate that despite of their low thermal conductivity, the present TMDs can exhibit large hydrodynamic effects, being comparable to those of graphene, especially for WSe_2 at high temperatures.

Keywords: thermal conductivity, transition metal dichalcogenides, first-principles, phonon hydrodynamics, heat transport, Normal scattering

1. Introduction

Since the isolation of monolayer graphene [1], research in 2D materials has experienced a significant increase [2]. Transition metal dichalcogenides (TMDs) [3, 4] have attracted a huge interest due to their particular physical properties, such as tunable thickness-dependent bandgap [5, 6] or anisotropic response under tensile strain [7]. Similarly to graphene, TMDs have covalent in-layer bonds and weaker van der Waals out-of-layer interactions and thus, beside being synthesized bottom-up, they can also be exfoliated, allowing the exploration of atomically thin layers [5]. They have a significant bandgap, which allows bypassing the most important limitation of graphene when it comes to the design and engineering of electronic devices [8]. Their bandgap is indirect and ranges from 0.85 to 1.23 eV in the bulk, while it increases up to 1.5-1.8 eV and it becomes direct for the monolayer, a fact that paves the way for their use in photonic applications [9, 10]. Additionally, they are good candidates for applications in catalysis, energy storage, and sensing, at variance with graphene, which is chemically inert unless functionalized with specific molecules [11].

The thermal properties of TMDs have been comparatively less studied, but they are attracting a growing interest [12]. A few theoretical and experimental works have started to determine the properties of TMDs with high accuracy. The phonon dispersions were the first features to be studied from a theoretical viewpoint [13, 14, 15] and the Raman active modes were also experimentally reported [15]. Further theoretical calculations have been able to determine the thermal conductivity from molecular dynamics [16, 17, 18] and from lattice dynamics using density functional theory (DFT) through the Boltzmann transport equation (BTE) under the relaxation time approximation (RTA) [19]. Recently, more accurate solutions of the BTE [20] have allowed to compute with greater accuracy the thermal conductivity of MoS₂ [21, 22].

In this work we present a unified, *ab initio* description of the thermal conductivity, κ , of single-layer transition metal dichalcogenides, MX₂ (M = Mo, W; X = S, Se) by obtaining the interatomic force constants from DFT calculations and then solving numerically the BTE beyond the RTA. We show that the latter can result in underprediction of κ up to the 50%. Additionally, we demonstrate that momentum conserving phonon-phonon scattering, *i.e.* normal (*N*) processes, are the predominant three-phonon collision mechanisms. This can lead to phonon hydrodynamics [23, 24, 25], a fluid-like heat transport regime characterized by the appearance of phenomena like Poiseuille flow and second sound, and whose relevance in graphene and in other 2D materials has already been highlighted [21, 26].

2. Computational methods

Heat transport in dielectric materials can be described by the phonon BTE

$$\frac{\partial n_{\mathbf{q}}}{\partial t} + \mathbf{v}_{\mathbf{q}} \cdot \frac{\partial n_{\mathbf{q}}}{\partial \mathbf{r}} = \left(\frac{\partial n_{\mathbf{q}}}{\partial t} \right)_{\text{scattering}}, \quad (1)$$

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which states that a perturbed distribution of phonons $n_{\mathbf{q}}$ restores its equilibrium state $n_{\mathbf{q}}^0$, expressed by the Bose-Einstein distribution function, through scattering (collision) processes [27].

This complex equation can be solved through different methods. A common approximation was the RTA, due to its simplicity. This model has been particularly useful as long as the size of the samples was quite big and the experiments were performed under slow heating conditions. The improvement of manufacturing techniques and new electronic devices have allowed to reduce the sample size down to few nanometers and the heating times can be of the order of picoseconds. In these cases it has been proved that the RTA solution is often far from the real solution, leading to the proposal of more accurate solutions to the linearized BTE. In this work three of them are considered: the direct solution derived by Chaput [28] (L-BTE), the iterative solution (I-BTE) proposed by Ward *et.al.* [29], and the kinetic collective model (KCM) approach due to De Tomás *et.al.* [30, 31].

The thermal conductivity derived from the L-BTE solution can be expressed in its final form as:

$$\kappa^{\alpha\beta}(\omega) = \int \frac{\rho_{\alpha\beta}(\omega')}{\omega' - i\omega} d\omega', \quad (2)$$

where $\rho_{\alpha\beta}(\omega')$ is a spectral density. For more details of the derivation of this solution we refer to the original article [28].

In the case of the I-BTE approach, an iterative process is done starting from the RTA solution in order to converge to the final solution. The thermal conductivity is then expressed as:

$$\kappa_{ij} = \frac{1}{k_b T^2 \Omega N} \sum_{\mathbf{q}} n_{\mathbf{q}}(n_{\mathbf{q}} + 1) (\hbar \omega_{\mathbf{q}})^2 v_{i,\mathbf{q}} \tau_{\mathbf{q}} (v_{j,\mathbf{q}} + \Delta_{j,\mathbf{q}}), \quad (3)$$

where Ω is the volume of the unit cell, N the number of \mathbf{q} -points used in the sampling, $n_{\mathbf{q}}$ the Bose-Einstein distribution function, $\omega_{\mathbf{q}}$, $v_{\mathbf{q}}$ and $\tau_{\mathbf{q}}$ correspond to the phonon frequency, velocity and relaxation time respectively. The term $\Delta_{j,\mathbf{q}}$ captures the deviation on the heat current with respect to the RTA approach. In highly kinetic materials like alloys, this difference between both approaches is close to zero, while it can be very significant in the materials we are studying here.

In order to evaluate more precisely how important this kinetic contribution is with respect to the real solution, the KCM was recently proposed [30, 31]. Here the total contribution to the thermal conductivity is split into a kinetic and a collective term, where the former corresponds to the effect of individual phonon collisions and the latter to a global contribution due to the coupling of phonon modes as a result of N processes. This allows to evaluate the collective contribution to κ in materials with relevant momentum conserving scattering processes:

$$\kappa_{ij} = \kappa_{ij}^K (1 - \Sigma) + \kappa_{ij}^C \Sigma, \quad (4)$$

where κ_{ij}^K corresponds to the kinetic contribution to the thermal conductivity and κ_{ij}^C to the collective one. The Σ factor is a weighting term that determines the predominance

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of each term depending on the dominance of N versus resistive (R) phonon-phonon scattering processes $\Sigma = \Gamma_N/(\Gamma_N + \Gamma_R)$. The resistive collisions are those that contribute directly to the thermal resistance, such as Umklapp (U) and impurity/mass defect scattering. The interested reader can find expressions for κ_{ij}^K and κ_{ij}^C in Ref. [31].

All the previous discussed solutions of the BTE are currently implemented in open-source programs. The L-BTE can be solved by using the Phono3py package [32], the I-BTE is implemented in the ShengBTE software [20] and the KCM in the KCM.PY code [33].

In order to determine the thermal conductivity of the present TMDs it is necessary to calculate their harmonic and anharmonic properties. To this end we have carried out DFT calculations using the VASP code [34] with the Local Density Approximation (LDA) for the exchange-correlation energy functional, as parameterized by Perdew and Zunger [35] from Ceperley and Alder's quantum Monte Carlo results [36]. We have used a plane wave cutoff ranging from 279 to 323 eV, depending on the material studied, with the projector augmented-wave (PAW) method [37, 38]. Born effective charges have been considered to include the non-analytic corrections of the LO-TO bands at the Γ point. The single-layer TMDs are separated by a vacuum buffer layer of approximately 6 Å from their periodic images; we checked that this vacuum thickness yields converged phonon dispersions compared to the results of Molina-Sánchez and Wirtz [13] who used 12 Å. For the harmonic and anharmonic properties, a $6 \times 6 \times 1$ supercell with a $4 \times 4 \times 1$ \mathbf{k} -point sampling was used. The second and third order interatomic force constants (IFC) have been calculated by the finite displacement method. The ShengBTE and Phono3py codes use the harmonic IFCs calculated from Phonopy [39]. Here interactions between all the atoms of the supercell are taken into account. To generate the displacements for the anharmonic IFCs, ShengBTE requires an external program called THIRDORDER.PY, while Phono3py has its own script inside the code. Regarding KCM.PY, it uses the output of Phono3py in order to calculate the thermal conductivity under its approach.

The details of the number of neighbors considered in the 3-phonon scattering processes and the \mathbf{q} -point sampling used to calculate the anharmonic properties and the thermal conductivity will be discussed in the next section.

3. Results and discussion

3.1. Convergence study

As an initial step of our analysis, we have conducted a detailed convergence study of the IFCs with respect to a few important parameters. We have studied (i) the supercell (SC) size used for the calculation, (ii) the number of neighbors considered in the anharmonic processes, (iii) the number of \mathbf{q} -points used to sample the Brillouin zone, and (iv) the Gaussian smearing factor. Carefully converging all these parameters, as we discuss below, is essential to obtain a reliable estimate of the thermal conductivity and to compare the predictions of the different approaches to solve the BTE.

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3.1.1. Supercell size Two dimensional materials such as graphene or the single-layer TMDs here studied feature a flexural acoustic phonon band that has a quadratic dispersion (*i.e.* zero sound velocity) close to Γ . Preliminary calculations showed that it is quite challenging to describe well the curvature of this phonon band as it approaches Γ in TMDs, and that the results critically depend on the SC size. Therefore, we took this as a criterion to establish which is the minimum cell size required to have well converged harmonic IFCs. Anharmonic IFCs typically converge faster than harmonic ones and it is common to use slightly smaller SCs for the latter [20, 40]. Indeed, we observe that providing a satisfactory description of the flexural phonon band turned out to be a rather stringent requirement that can be fulfilled only with relatively large cells (see Fig. 1). As a result of this convergence test we decided to use a $6 \times 6 \times 1$ SC.

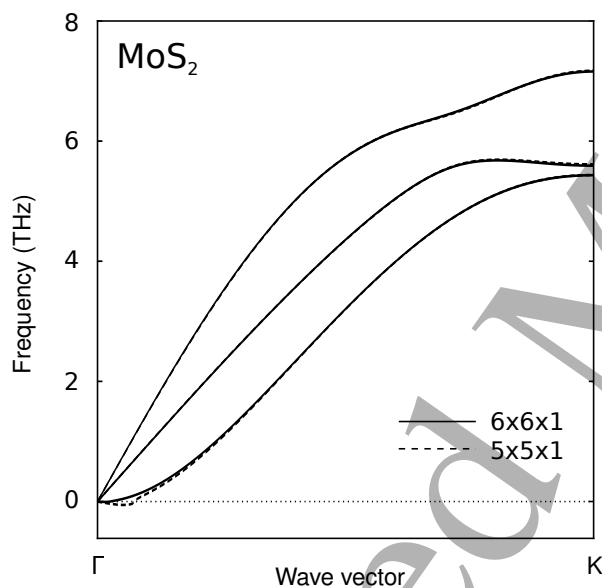


Figure 1. Acoustic phonon bands for MoS₂ along the $\Gamma - K$ direction using a $5 \times 5 \times 1$ and a $6 \times 6 \times 1$ supercell.

3.1.2. Cutoff for three-phonon processes We used a $6 \times 6 \times 1$ SC also for the calculation of the anharmonic IFCs by finite displacements. Computing them requires, in principle, to carry out a few hundreds of DFT calculations for each material. To reduce the number of displacements considered, and given the short range of the non-dipole contribution to the anharmonic IFCs [41, 42], it is sometimes possible to calculate three-phonon collision processes neglecting interactions between atoms beyond a given cutoff. Small cutoffs result in less atom triplets to study and thus a lighter computational load, but also a lower accuracy. Therefore, we have studied the convergence of the thermal conductivity of MoS₂ considering interactions only up to n -th neighbors and increasing n .

For this convergence study we settled on the L-BTE solution implemented in Phono3py because the results should not depend on how the BTE is solved, but rather

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on the range of the anharmonic IFCs within the DFT computational framework adopted. Additionally, Phono3py allows generating the complete sets of displacements, up to all neighbors, while the cutoff can later be independently specified when solving the BTE. Conversely, this is not possible with THIRDORDER.PY as it is not straightforward to extract from the set of displacements generated for a large number of neighbors the set for a smaller number, as the indexing of the displacements changes.

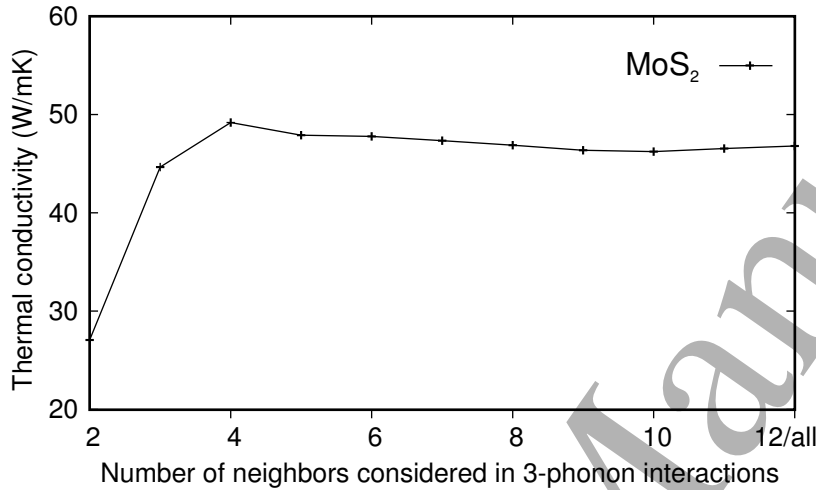


Figure 2. Thermal conductivity convergence with an increasing number of neighbors considered for three-phonon scattering processes in MoS₂, using the full solution of the L-BTE from Phono3py [32].

Our results are summarized in Fig. 2, where we plot the thermal conductivity of MoS₂ as a function of the number of neighbors. For these calculations we have used a $60 \times 60 \times 1$ \mathbf{q} -point mesh sampling, which, as shown below, is well converged. For the $6 \times 6 \times 1$ SC adopted, the maximum number of neighbors that can be considered is 12. Thus, this last value corresponds to a calculation without any limitation on the number of atoms in the three-phonon scattering processes, *i.e.* no cutoff. It can be seen that after 5 neighbors the variations on the thermal conductivity are very small. We therefore considered interactions up to 6th neighbors for the calculation of the anharmonic IFCs.

3.1.3. Brillouin zone sampling Once the harmonic and anharmonic IFCs are properly calculated, following the convergence prescriptions described in Sections 3.1.1 and 3.1.2, the BTE must be solved on a grid of \mathbf{q} -points that samples the Brillouin zone. Therefore, the next step of our preliminary convergence study is to check the dependence of the thermal conductivity with respect to the number $N_{\mathbf{q}}$ of \mathbf{q} -points in an $N_{\mathbf{q}} \times N_{\mathbf{q}} \times 1$ grid. We perform this test at room temperature, and our results are shown in Fig. 3. Note that in this case, as we are dealing with the way the BTE is solved, we carried out the convergence test with the three approaches considered, *i.e.* L-BTE, I-BTE, and KCM. As seen in Fig. 3, a $60 \times 60 \times 1$ mesh provides converged results in all cases, and this is the mesh we took for all systems and models. Here we have limited the phonon

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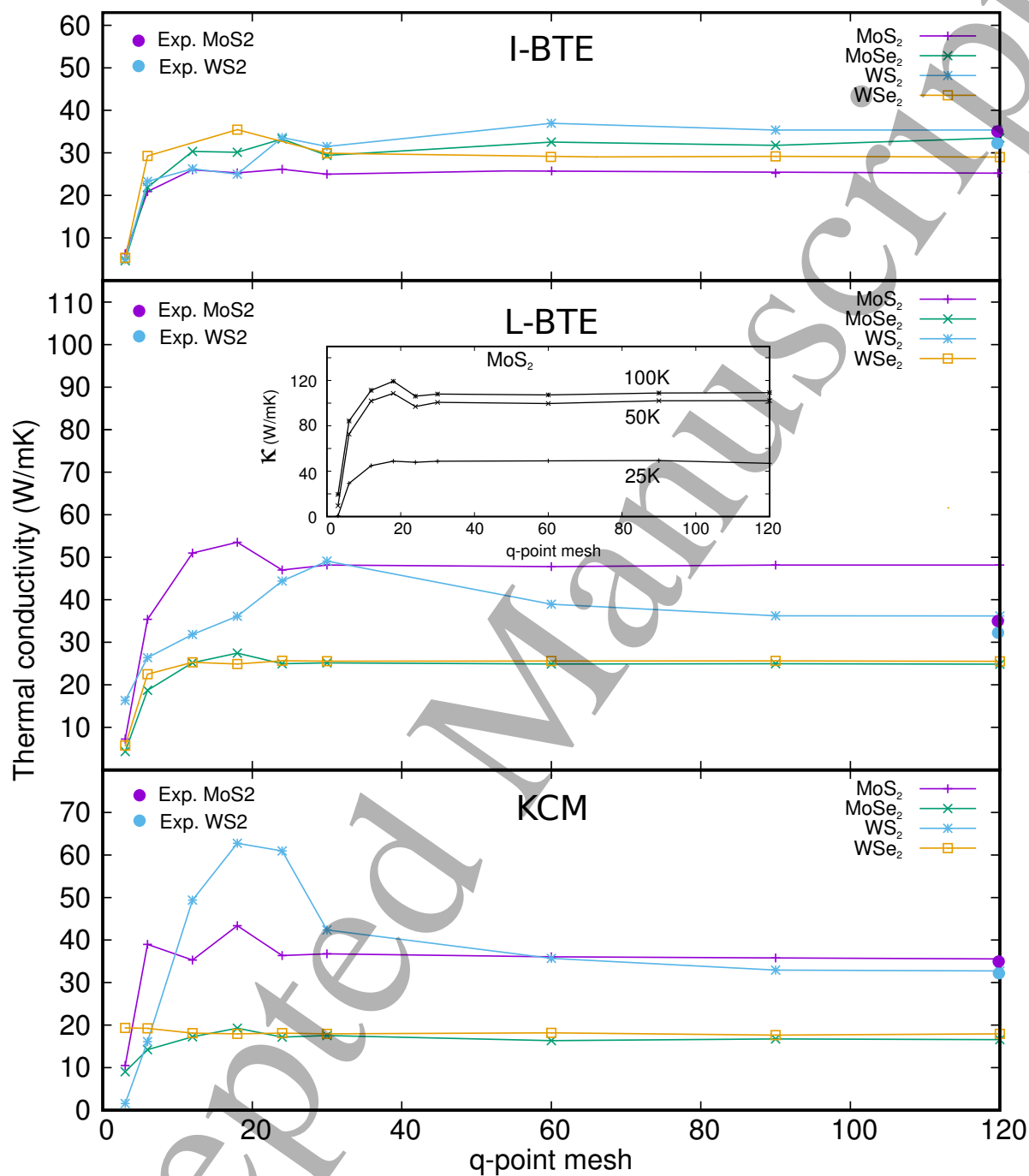


Figure 3. Thermal conductivity convergence with respect to the \mathbf{q} -point mesh for the I-BTE, L-BTE and KCM.

mean free path (MFP) to $1 \mu\text{m}$ in order to work with a sample size of the order of the available experimental data [43, 44], reducing at the same time the computational cost. The value of $N_{\mathbf{q}}$ needed to converge the thermal conductivity depends on the temperature and at lower temperatures, where long wavelength phonons are important for heat transport, finer \mathbf{q} -point grids might in principle be required. In the inset of Fig. 3 we plot κ of MoS_2 calculated with L-BTE as a function of $N_{\mathbf{q}}$ at 25, 50 and 100 K.

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It can be observed that the limitation of $1 \mu\text{m}$ also ensures convergence with a $60 \times 60 \times 1$ grid at these lower temperatures as well. Notice, however, that these conclusions should be revised for boundary scattering conditions corresponding to larger samples, where finer \mathbf{q} -meshes might be needed, particularly at low temperatures.

3.1.4. Gaussian smearing factor Another tunable parameter that governs the thermal conductivity is the smearing factor used to define the Gaussian function that modulates energy and momentum conservation in the three-phonon scattering processes. For the I-BTE solution it has been checked that the iterative process only converges in all cases for $\sigma \leq 1 \text{ cm}^{-1}$ for \mathbf{q} -point meshes up to $60 \times 60 \times 1$. The need to reduce σ to converge the solution of the I-BTE was previously reported in the study of high-pressure polymorphs of silica [45]. In the case of the L-BTE, the solution is not that sensitive and it converges up to $\sigma \leq 10 \text{ cm}^{-1}$. Therefore we have used these two values for each model. In the case of the KCM, its solution will converge in all cases and thus we have used the same values as in the L-BTE solution, as KCM.PY uses its output.

3.2. Thermal conductivity

The harmonic properties of the four TMDs investigated are shown in Figure 4. Despite an overall similar shape of the dispersion relations there are some quantitative differences. The most important one concerns the sound velocities of Se-based materials, which are lower than those of S-based materials. As discussed below this plays an important role in the determination of κ . Interestingly, MoSe_2 exhibits some difference from the common pattern of the other materials: it has a very small acoustic-optical gap and its optical phonons are split into two subbands. However, these features do not result in a measurable difference in the thermal conductivity, and $\kappa(\omega)$ of MoSe_2 and WSe_2 are at a first approximation indistinguishable (see below).

The thermal conductivities as a function of temperature are shown in Figure 5. As it can be seen, the different numerical solutions of the BTE provide estimates of κ that can differ from each other. As a general rule the iterative solution of the BTE yields lower values for κ than those obtained from the L-BTE. Nevertheless, while these differences are negligible in the case of WS_2 and are within $2\text{-}5 \text{ W m}^{-1} \text{ K}^{-1}$ at 300 K for MoSe_2 and WSe_2 , they are very large for MoS_2 . The experimental room temperature value of κ for WS_2 is well predicted by all methods, while the value for MoS_2 falls in between the results obtained from I-BTE and L-BTE. The most accurate prediction for MoS_2 is provided by the KCM approach. As a possible cause of the overestimation provided by the L-BTE for MoS_2 , we note that it has been recently found that four-phonon scattering can reduce the intrinsic thermal conductivity of solids [46]. This is achieved by four-phonon processes substantially reducing the lifetime of optical phonons in materials with large acoustic-optical (a-o) gap. Given that, among the four materials studied, MoS_2 has both the largest contribution from optical phonons and the largest a-o gap, it is conceivable that MoS_2 is the material most sensitive to four-phonon processes.

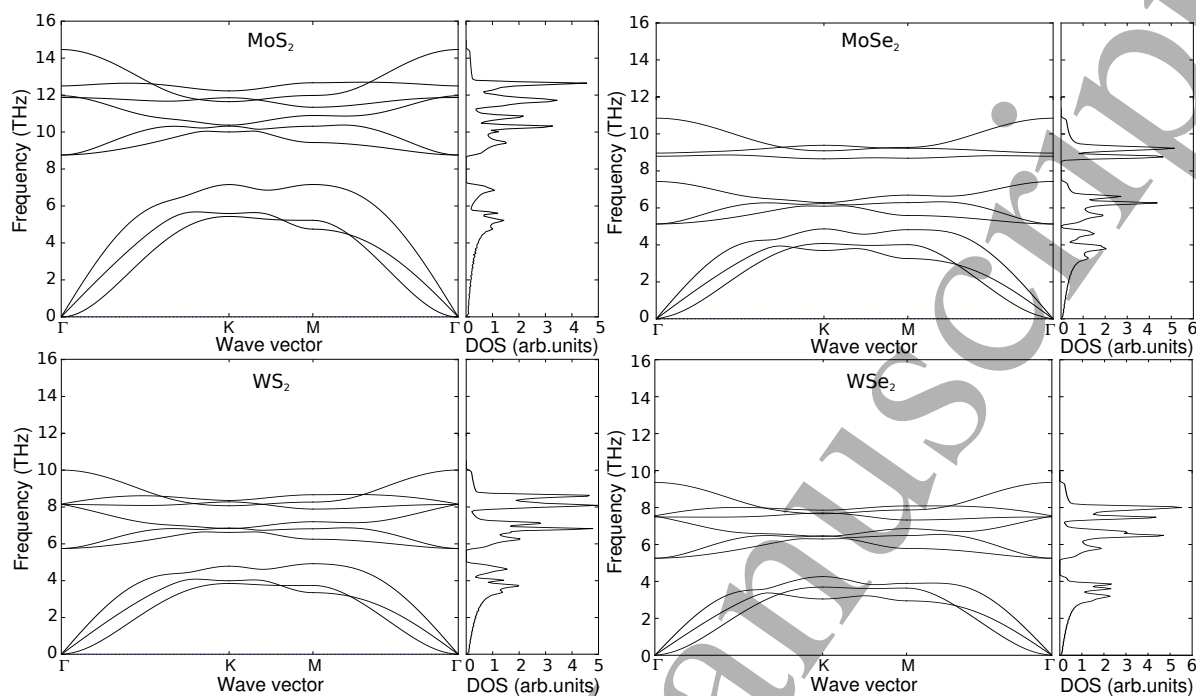


Figure 4. Phonon dispersion and density of states (DOS) of MoS₂, MoSe₂, WS₂ and WSe₂.

On the other hand, the underestimation of κ as obtained from the iterative solution of the BTE may depend, as previously discussed, on the use of the RTA solution as initial guess. In materials with low atomic masses where N scattering plays an important role, the RTA solution can be sufficiently far from the exact solution and the iterative process might not converge [47]. In the present case, for instance, the RTA solution for MoS₂ is only 33.1 % of the experimental value, while for WS₂ is 47.2 %. From Figure 5 it can be also observed that the RTA solution behaves different for S- and Se-compounds. The underestimation for S-based materials is higher due to the lighter weight of the sulphur atoms (less than half of selenium), in agreement with previous results for diamond [29, 31] and graphene [21, 48], where the low weight of the carbon atoms and the importance of N scattering makes the RTA approach underestimate the thermal conductivity. A similar disagreement between I-BTE and L-BTE has previously been observed also in GaN [49, 50, 51].

To gain a better insight on which parts of the phonon spectrum contribute more to the thermal conductivity at room temperature and to allow an easier quantitative comparison among the different materials, we have calculated the participation ratio, i.e. the modal decomposition $\kappa(\omega)$, which is shown in the insets of Figure 5. As it can be seen there, the thermal conductivity is almost entirely determined by the acoustic modes. This contribution is larger for S-based materials, partly because of the higher phonon velocities, and thus results in larger values of κ . The accumulation function of the participation ratios plotted in Figure 6 clearly shows that there are two groups of materials: S-based compounds with a κ of $\sim 35 \text{ W m}^{-1} \text{ K}^{-1}$ and Se-based compounds

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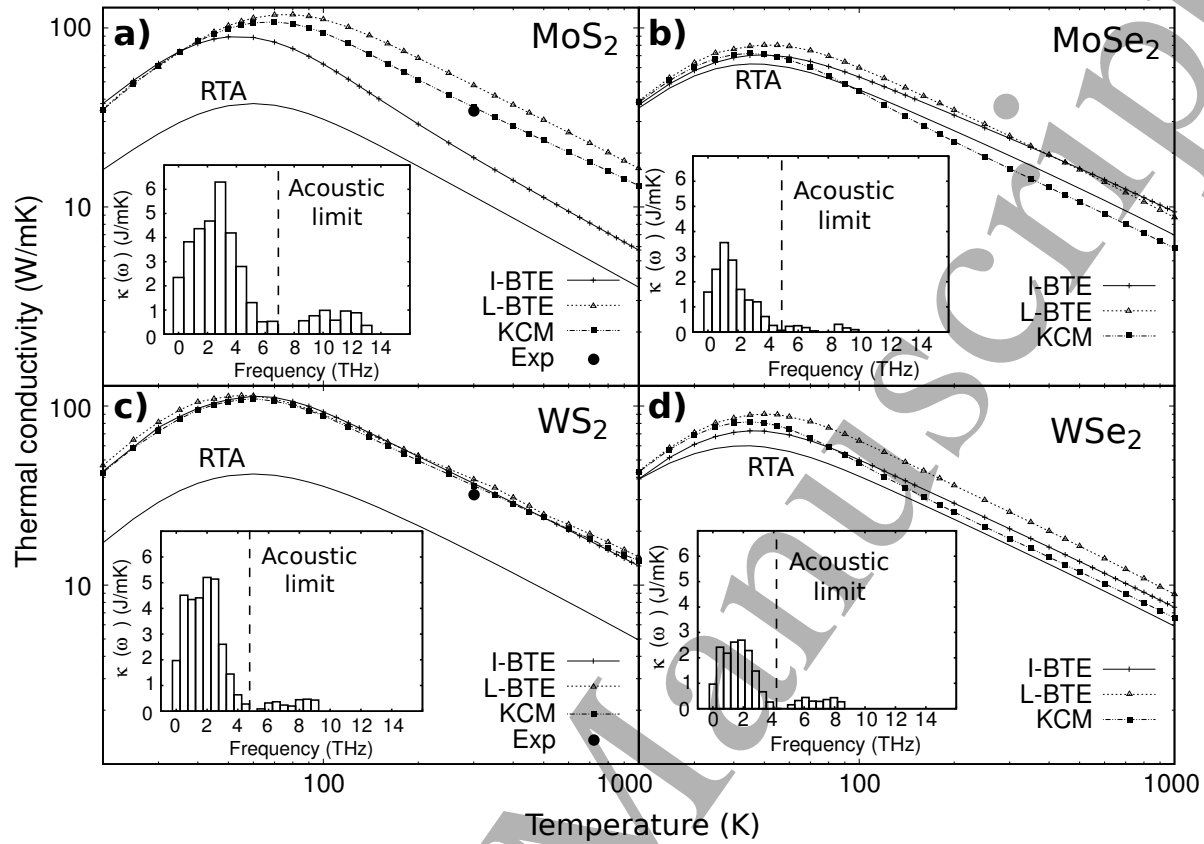


Figure 5. Thermal conductivity as a function of temperature of single-layer (a) MoS₂, (b) MoSe₂, (c) WS₂ and (d) WSe₂ using the I-BTE, L-BTE and KCM models. The RTA solution is also included for comparison. The experimental room-temperature values for MoS₂ (Ref. [43]) and WS₂ (Ref. [44]) are also shown. The participation ratios, *i.e.* the frequency resolved thermal conductivity $\kappa(\omega)$, at 300 K using the L-BTE model are plotted in the insets.

with a κ of $\sim 15 \text{ W m}^{-1} \text{ K}^{-1}$. MoS₂ is the only material where a non negligible contribution to κ comes from optical phonons with frequency between 8 and 12 THz, a feature that can also be seen in the corresponding inset in Figure 5. The similarity between WSe₂ and MoSe₂ is noteworthy, not only for the value of κ , but also for its modal decomposition.

3.3. Phonon hydrodynamics

In thermal transport, a hydrodynamic heat transport regime is expected when momentum conservation in the phonon distribution is important. This can happen, for example, when momentum conserving collisions are dominant when compared to resistive (R) processes [52, 23]. The first group are the so-called normal (N) processes [53]. The second group includes the intrinsic collision processes that contribute directly to the thermal resistance, *i.e.*, Umklapp (U) and impurity/mass defect scattering (I). In order to determine the relaxation time associated to each scattering mechanism we take advantage of KCM.PY code [33], which provides the averaged values

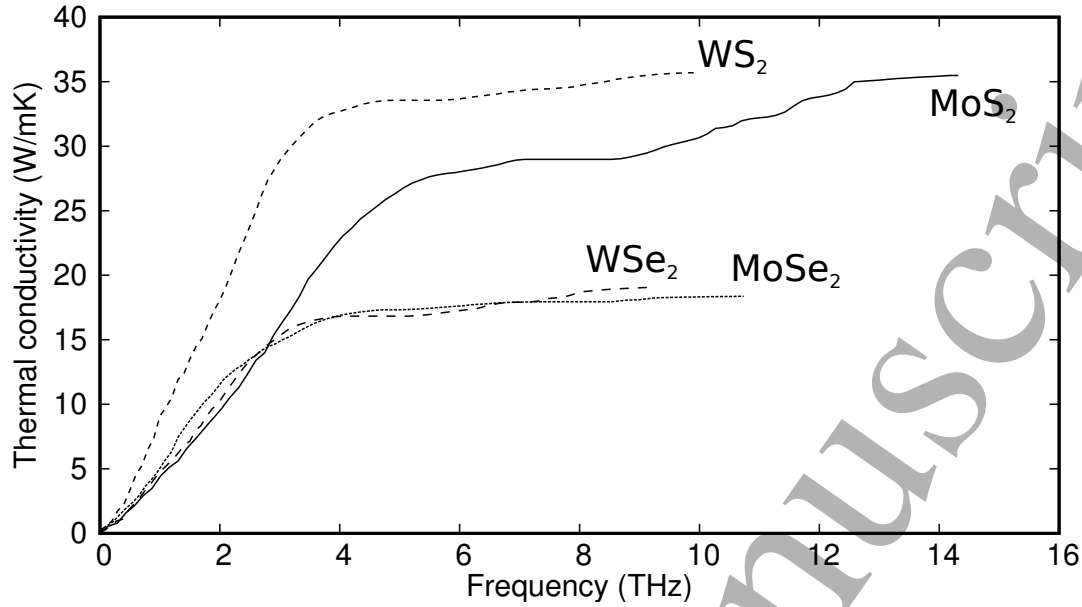


Figure 6. Accumulated participation ratios $\int_0^\omega \kappa(\omega') d\omega'$ at 300 K of MoS₂, MoSe₂, WS₂ and WSe₂ using the L-BTE model.

of these phonon scattering relaxation times [31] as a function of temperature.

The relaxation times for momentum conserving collisions (N) and resistive processes ($U+I$) are represented in Fig. 7. From this representation it can be observed that for all materials the trend of both processes as function of the temperature is quite similar. While the relaxation times are almost constant at high temperatures, as the temperature decreases phonons recover their equilibrium state slower, i.e. longer relaxation times. Here it is important to notice that in all cases there are almost two orders of magnitude of difference between the R and N processes, that means that N processes dominate heat transport at all temperatures. The N and R relaxation times for graphene are also represented for comparison, showing a very similar trend. It can be also observed that at low temperatures the differences between N and R scattering for the TMDs are smaller, as the I scattering becomes relevant. For graphene, due to the lower isotopic abundance this is not observed. These results are in agreement with previous calculations of graphene and MoS₂ [21].

In recent articles it has been discussed that the dominance of N collisions, as observed in Fig. 7, can be the origin of collective effects and hydrodynamic heat transport [30, 21, 31, 54]. Therefore we analyze the contribution of this regime to the thermal conductivity through the splitting of the thermal conductivity proposed by the kinetic-collective approach [30, 31].

In Fig. 8 the collective contribution to thermal transport, $\kappa^C \Sigma / \kappa$, is represented. For all materials and at all temperatures, the collective contribution to the thermal conductivity is very significant. It can be observed that all the TMDs show a very similar contribution in all the temperature range, being a 20-30 % at high temperatures. In a range of temperatures between 25-60 K, impurity scattering corresponding to

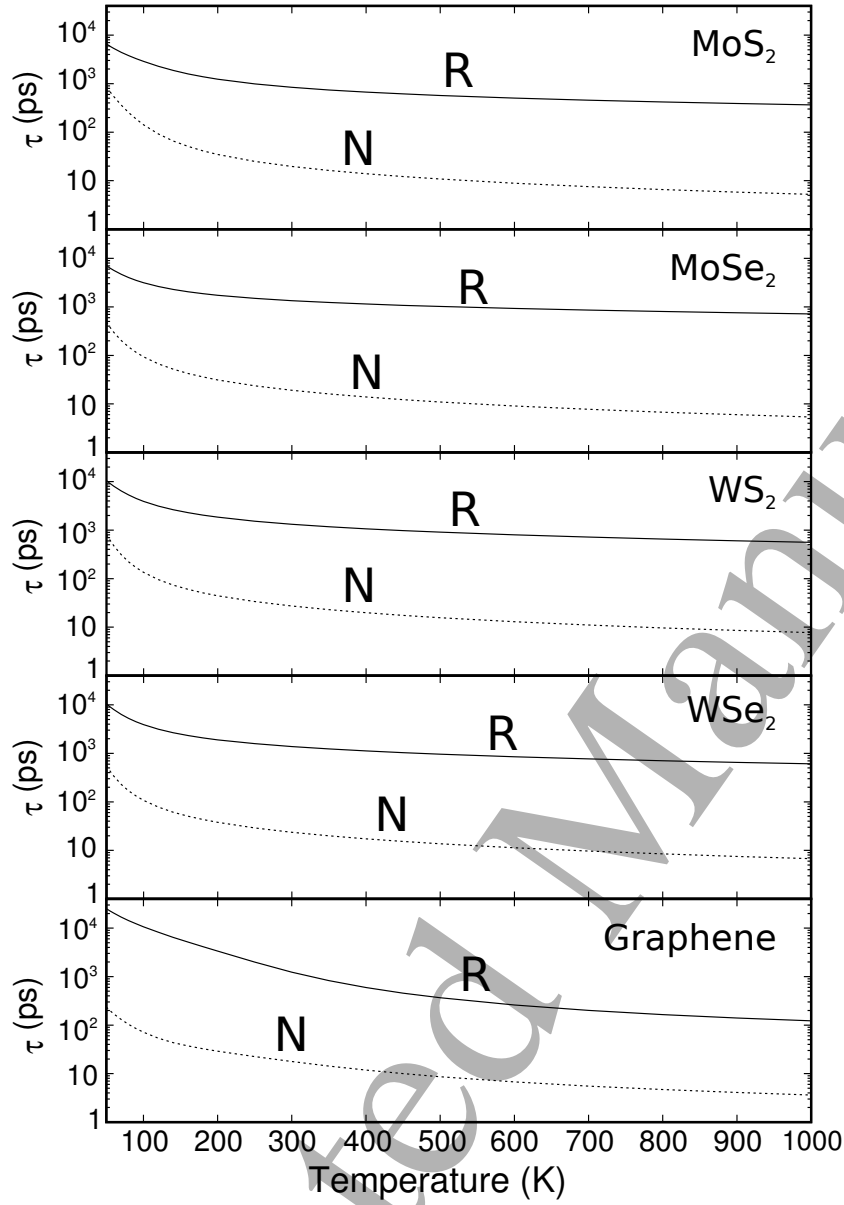


Figure 7. Normal (N) and resistive (R) including Umklapp (U) and impurity (I) relaxation times for MoS₂, MoSe₂, WS₂ and WSe₂. Graphene is also included for comparison.

natural isotopic abundances dominates the thermal transport, producing a decrease of the collective contribution to κ . This decrease is sharper in the case of MoS₂ and WS₂. As expected for this kind of 2D materials, and due to the dominance of N processes, especially at very low temperatures, the collective contribution increases as the temperature goes down below 25 K (when impurity effects vanish), achieving values as high as 40 % at 20 K and tending to 100 % when $T \rightarrow 0$. The collective contribution to κ of graphene is quite different. While at high temperatures it is around 20 %, the impurity scattering has a negligible effect allowing to N scattering dominating the thermal transport as the temperature goes down. Notice that at room temperature such

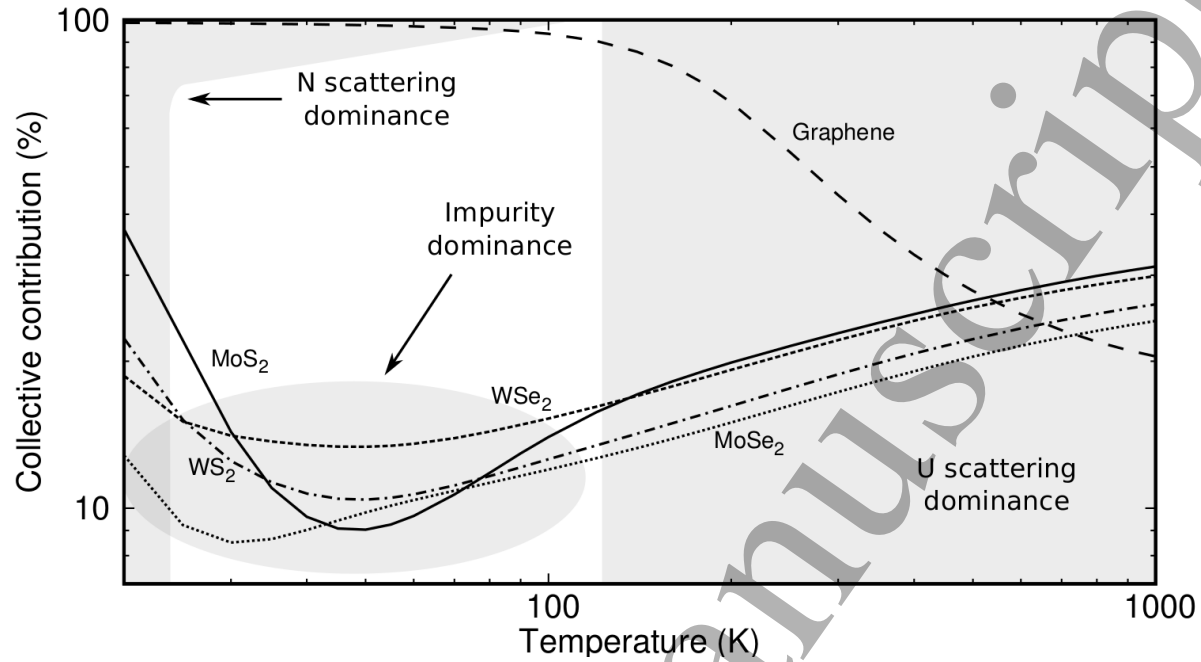


Figure 8. Collective contribution to the total thermal conductivity according to the KCM for the four studied single-layer TMDs as well as for graphene.

contribution is around 40 % of the total κ , and below 90 K it is more than 90 %.

Another parameter that has been useful to explain recent experiments in a hydrodynamic framework is the so-called non-local length [23, 24, 54, 55]. By solving the BTE [Eq. (1)] in a moment basis, one can obtain a hydrodynamic-like equation connecting the heat flux \mathbf{Q} to the thermal conductivity κ through a non-local length ℓ related to the heat viscosity of a certain material [23, 24]:

$$\mathbf{Q} - \ell^2 (\nabla^2 \mathbf{Q} + 2\nabla \nabla \cdot \mathbf{Q}) = -\kappa \nabla T. \quad (5)$$

Originally this equation was derived by Guyer and Krumhansl [23, 24] for a pure collective regime in which $N \gg R$, leading thus to a collective non-local length ℓ_C . In more recent works, this parameter has been generalized to a first order to explain experiments where $N \sim R$ [54, 55], leading to a kinetic contribution to the non-local length ℓ_K .

The collective non-local lengths for all the studied TMDs, as well as for graphene, are represented in Fig. 9. According to these values, hydrodynamic effects in a pure collective regime will be expected at scales between 10-100 nm at high temperatures, between 100 nm and 1 μm around room temperature and above 10 μm for $T \rightarrow 0$. Eventually, in cases where $N \sim R$ or even when $N < R$, hydrodynamic effects can be observed at larger length scales due to the kinetic contribution [54, 55]. In addition, to get the full picture, it should be noticed from Eq. (5) that hydrodynamic effects will be observable only when both the non-local length ℓ and the geometric term $(\nabla^2 \mathbf{Q} + 2\nabla \nabla \cdot \mathbf{Q})$ are important.

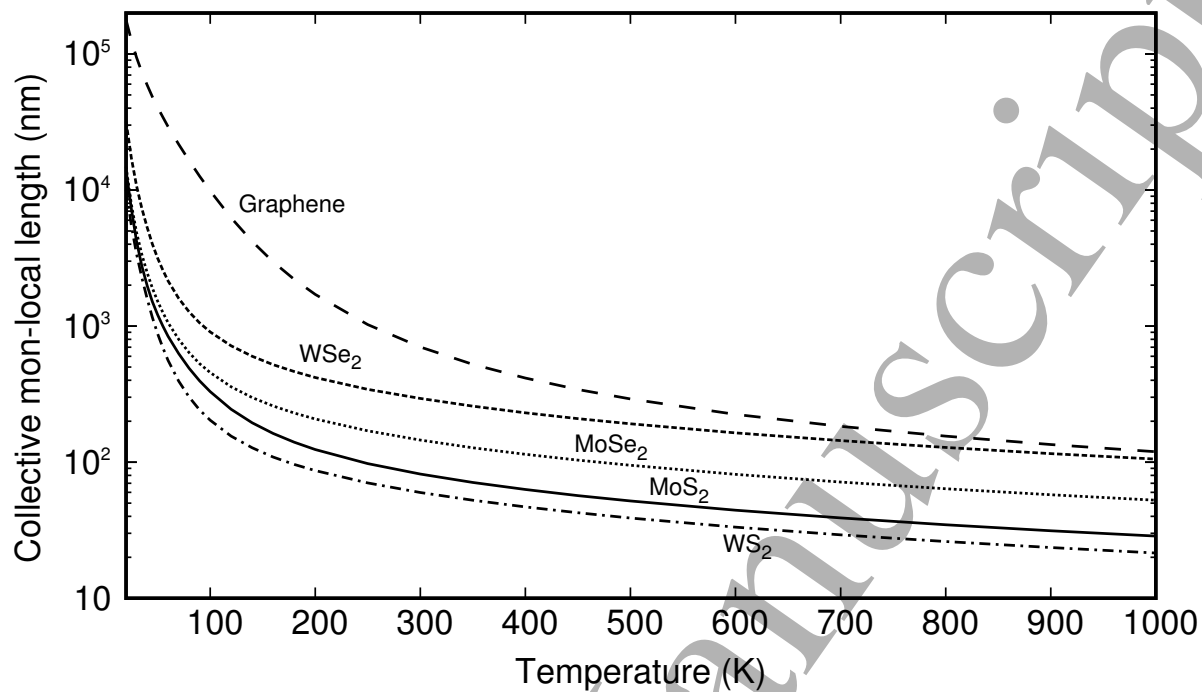


Figure 9. Collective non-local length (ℓ_C) as function of temperature for MoS₂, MoSe₂, WS₂ and WSe₂. The ℓ_C for graphene is also included for comparison.

4. Conclusions

Three different approaches to solve the BTE beyond the RTA have been used to study the thermal conductivity of four single-layer TMDs: MoS₂, MoSe₂, WS₂ and WSe₂. We have shown that $6 \times 6 \times 1$ supercells, anharmonic interactions up to 5th-6th neighbors and a \mathbf{q} -point grid of $60 \times 60 \times 1$ are necessary to obtain converged values of the thermal conductivity for a sample size of 1 μm . When available, the experimental data agree well with our results, although the iterative solution results in a non negligible underprediction in the case of MoS₂. Also, we see that the RTA significantly underpredicts the κ values for MoS₂ and WS₂. The temperature dependence is standard, where κ initially increases due to a higher number of phonons present, peaking in the 50-100 K range, and then decreasing as Umklapp phonon-phonon scattering takes over. Also, the spectral thermal conductivity indicates that almost all the contribution to the room temperature κ comes from acoustic phonons. In addition, we found a difference of more than one order of magnitude between the Normal and the Resistive relaxation times for all the materials investigated, a fact that could lead to the emergence of hydrodynamic effects similar to those already observed in graphene. WSe₂, in particular, appears to be the best candidate to observe such effects.

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