scientific reports



OPEN

Revealing the quantum nature of the voltage-induced conductance changes in oxygen engineered yttrium oxide-based RRAM devices

F. L. Aguirre^{1,2,6⊠}, E. Piros^{3,6⊠}, N. Kaiser³, T. Vogel³, S. Petzold³, J. Gehrunger⁴, C. Hochberger⁴, T. Oster⁵, K. Hofmann⁵, J. Suñé¹, E. Miranda¹ & L. Alff³

In this work, the quasi-analog to discrete transition occurring in the current–voltage characteristic of oxygen engineered yttrium oxide-based resistive random-access memory (RRAM) devices is investigated in detail. In particular, the focus of our research is not on the absolute conductance values of this characteristic but on the magnitude of its conductance changes occurring during the reset process of the device. It is found that the detected changes correspond to conductance values predominantly of the order of the quantum unit of conductance $G_0 = 2e^2/h$, where e is the electron charge and h the Planck constant. This feature is observed even at conductance levels far above G_0 , i.e. where electron transport is seemingly diffusive. It is also observed that such behavior is reproducible across devices comprising yttrium oxide layers with different oxygen concentrations and measured under different voltage sweep rates. While the oxygen deficiency affects the total number of quantized conductance states, the magnitude of the changes in conductance, close to 1 G_0 , is invariant to the oxygen content of the functional layer.

Resistive random-access memory (RRAM) is one of the most promising emerging non-volatile memory technologies as it is characterized by competitive performance such as high scalability (< 10 nm)¹, ultrafast (< 1 ns)² and low-power switching (1 pJ per bit)³, high endurance (>109 cycles)⁴ and good radiation hardness^{5,6}. RRAM devices rely on the resistive switching mechanism which has been observed in a number of transition metal oxides such as hafnium⁷, tantalum⁸, and yttrium⁹ oxide, all of which are considered as alternative gate oxide materials in complementary metal-oxide-semiconductor (CMOS) technology. As it has been reported several times, in valence change memories, the information storage relies on the formation and rupture of conductive filaments formed of oxygen vacancies. When the vacancy bridge connects both electrodes, the device switches to the low resistive state (LRS), whereas when the filament is ruptured at opposite bias, the high resistive state (HRS) is recovered. According to mesoscopic theory¹⁰, when the cross-section of the conductive filament is comparable to the electronic wavelength, quantum effects are expected to arise in the system even at room temperature. Under this circumstance, the constriction's bottleneck behaves as a quantum point contact (QPC) in which the electron transport exhibits quantized steps in terms of the fundamental unit of conductance (quantum conductance, $G_0 = 2e^2/h = (12.9 \text{ k}\Omega)^{-1}$ 11-13, where \hat{e} is the electron charge and \hat{h} the Planck constant). The stabilization of a large number of quantized conductance steps can be potentially utilized for (quantized) multi-bit storage 14,15. The resulting quasi-analog to discrete switching is also desired for neuromorphic applications that mimic the synaptic connectivity of biological neurons¹⁶.

¹Departament d'Enginyeria Electrònica, Universitat Autònoma de Barcelona, 08193 Cerdanyola del Valles, Spain. ²Intrinsic Semiconductor Technologies, Ltd., Buckinghamshire, United Kingdom. ³Advanced Thin Film Technology Division, Institute of Materials Science, Technische Universität Darmstadt, Darmstadt, Germany. ⁴Computer Systems Group, Department of Electrical Engineering and Information Technology, Technische Universität Darmstadt, Darmstadt, Germany. ⁵Integrated Electronic Systems Lab, Department of Electrical Engineering and Information Technology, Technische Universität Darmstadt, Darmstadt, Germany. ⁶These authors contributed equally: F. L. Aguirre and E. Piros. [™]email: aguirref@ieee.org; eszter.piros@tu-darmstadt.de

In the last years, yttrium oxide (Y₂O₃) has been gaining increasing interest as the active material in filamentary-type RRAMs because of the intrinsically vacant anion sublattice sites occurring even in the stoichiometric compound, resulting in a universal low-frequency noise and good thermal stability¹⁶⁻¹⁹. The intrinsic oxygen vacancies are reported to form chains along the [110] directions within the cubic phase that can serve as preferential paths for the filament formation, thus contributing to the low electroforming and operation voltages^{9,20,21}. It has also been shown¹⁶ that by engineering the oxygen content, a digital-to-(quasi-)analog transition can be induced in the switching characteristics: by increasing the degree of oxygen deficiency a high number of stable intermediate resistive states are created during both the set and reset processes. Moreover, these intermediate levels can be described in terms of integer and non-integer values of G_0 , i.e., they display nonlinear conductance quantization as understood in the framework of the generalized mesoscopic transport model¹⁶. In this work, we do not focus on the quantized conductance levels themselves, but specifically on the magnitude of the transitions between intermediate states occurring during the reset process. The important result is that oxygen deficient Y_2O_x not only presents a multi-step or even fully analog/gradual switching, but that the transition from one level to the next occurs mostly in steps of $\sim 1~G_0$. This phenomenon occurs regardless of the oxygen content in the oxide layer, as the stoichiometry of the functional layer affects mostly the total number of transitions observed but not their magnitude. This indicates that the observed quantum behavior of the electron transport is a universal property in filamentary structures revealed when considering conductance *changes* instead of absolute values.

Results and discussion

The experiments shown in this work were conducted on RRAM devices consisting of a Si/Al/TiN/Y $_2$ O $_{3-x}$ /Pt structure. The functional layer was deposited using a molecular beam epitaxy setup that allows for controlling the oxygen stoichiometry of the yttria film by varying the flow of oxygen radicals (in total 5 different samples grown with -from least to most oxidized- 0.1, 0.2, 0.3, 0.5, and 1 standard cubic centimeter per minute, sccm) and the film growth rate (1 angstrom per second, Å s⁻¹, for all samples except for the most oxidized one grown with 0.25 Å s⁻¹). The electrode area is $30 \times 30 \ \mu m^2$ and the oxide thickness is 18.9 nm for the 0.1 sccm case, 16.8 nm for 0.2 sccm, 15.9 nm for 0.3 sccm, 15 nm for 0.5 sccm and 13.2 nm for 1 sccm. In all cases the exact thickness values have been derived from the period of the measured X-ray reflectometry (XRR) oscillations (using an RCRefSim software ²²). The yttria layers grown with equal to or higher than 0.3 sccm oxygen flow are characterized by a mixture between the monoclinic and cubic phases, whereas for the most oxygen deficient films (0.1 and 0.2 sccm) only a cubic phase is observed. All yttrium oxide layers are polycrystalline. The fabrication process and the complete structural characterization of the samples were described elsewhere ¹⁶. The electrical characterization was carried out using a Keithley 4200 semiconductor characterization system (SCS), biasing the Pt top electrode and grounding the TiN bottom electrode. The internal current compliance of the SCS was used in order to avoid the hard breakdown of the oxide layer during electroforming and set operations.

As previously reported¹⁶, oxygen engineering in yttria films allows to control the density of oxygen vacancies in the functional layer, which in turn enables the control over the dynamics of the filament formation and rupture. In highly oxidized yttria grown with 1 sccm oxygen flow, the switching is characterized by an abrupt transition. By decreasing the quantity of oxygen in the film, this behavior can be transitioned to a multilevel (at moderate oxidation, 0.5 sccm) and eventually, with further reduction of the oxygen content (below 0.5 sccm), to a quasi-analog switching characteristic both for the set and reset processes. This is illustrated in Fig. 1a.

To statistically study the distribution of the conductance variation in terms of G_0 , the reset transition was analyzed for 100 successive current–voltage (I-V) loops (see Fig. 1a). The obtained results are in agreement with previous works on this subject^{12,15,23-27}. Data from the reset region were converted into conductance-voltage (G-V), further referred as $g_i(v)$ with i identifying the loop number) and post-processed with a moving median filter²⁸ \mathcal{F}_{MM} to obtain a signal $g_{i,1}(v) = \mathcal{F}_{MM} \{g_{i,1}(v)\}$ free of noise but preserving the sharp edges corresponding

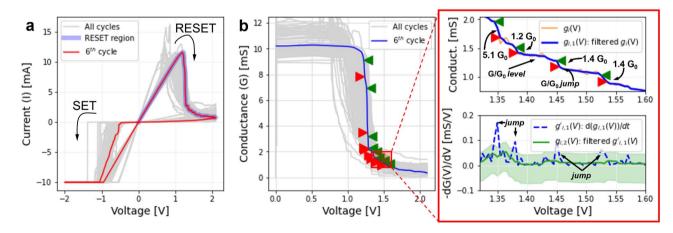


Figure 1. (a) I–V loops corresponding to the least oxidized sample (0.1 sccm oxygen flow during growth). The region (reset transition) used to extract the information about the transitions is highlighted in blue. (b) G-V representation of the reset transition. The green and red markers identify the starting and ending point of each jump. The zoom-in-panel at the right side of figure (b) shows a detail of the identified jumps (top) and how they are assessed using the derivative of the signal (bottom).

to the transitions from one conductance level to another. The resulting $g_{i,1}(\nu)$ curve is presented in Fig. 1b, and its derivative computed $\left(g'_{i,1}(\nu) = \frac{d(g_{i,1}(\nu))}{d\nu}\right)$. By further filtering $g'_{i,1}(\nu)$ using a moving average filter²⁸ \mathcal{F}_{MA} to obtain a reference signal $g_{i,2}(\nu) = \mathcal{F}_{MA}\left\{g'_{i,1}(\nu)\right\}$, it is possible to systematically identify the transitions between two different conductance levels by comparing $g'_{i,1}(\nu)$ against $g_{i,2}(\nu) + \delta$, δ being a threshold defining the minimal acceptable value of a conductance transition to be recognized as such, which in this case is 0.2 G_0 (see the green shaded region around $g_{i,2}(\nu)$ in Fig. 1b). The zoomed region in Fig. 1b exemplifies the procedure by presenting in the top axis a superposition of the signals $g'_{i,1}(\nu)$ and $g_{i,2}(\nu)$ with the tolerance region defined by $g_{i,2}(\nu) + \delta$. The transition conductance levels are identified by the points in which $g'_{i,1}(\nu) > g_{i,2}(\nu) + \delta$ and they are indicated in the bottom chart of the zoomed region by the green (start of the transition) and red (end of the transition) markers, which shows $g_{i,1}(\nu)$ for the same voltage region. Other examples are reported in Supplementary Fig. 1 to complement this assertion.

The method described above was used to locate the transitions between conductance levels on the five different oxygen engineered yttria devices investigated. The resulting histograms for the normalized conductance steps $(\Delta G/G_0)$ are shown in Fig. 2a-e. The insets show the I-V loops for the sample under test. The observed cycle-to-cycle variability is a result of morphological changes in the structure of the conducting pathway^{29,30}. For oxygen flows above 0.1 sccm (Fig. 2b-e), the number of transitions between conductance levels is found to be significantly lower during the set process, which is in line with the set transitions becoming increasingly abrupt when increasing the oxygen content. A similar behavior is observed in the reset transition, in which, despite showing some conductance transitions, the total count decreases for higher oxygen flows. The key finding is that the transitions between conductance levels during the reset process mostly occur in steps of $\sim 1~G_0$, this being independent of the degree of oxygen deficiency. In fact, for the case of the set process observed in the sample presented in Fig. 2a (oxygen flow of 0.1 sccm and growth rate of 1 Å s⁻¹) which also shows quantized conductance states, the transitions between them mostly occur in steps of ~ 1 G_0 . To confirm the findings reported in Fig. 2a–e, the identification of the transitions between conductance levels was repeated in other two devices fabricated in the same way as the one reported in Fig. 2a but varying the voltage sweep ramp rate from ~135 mV/s (Fig. 2f) to ~ 45 mV/s (Fig. 2h) and to ~ 35 mV/s (Fig. 2j). When probing conductance quantization with the voltage-sweep method, it is crucial to use a voltage-step size small enough to probe quantum conductance steps with the highest possible resolution and to keep simultaneously the overall measurement time at a reasonable level¹³. The joint assessment of Fig. 2f,h,j indicates that regardless of the voltage sweep rate, the general trend remains the same, that is, transitions between successive conductance levels are in the vast majority of ~ 1 G_0 i.e. the conductance value expected for an ideal monomode ballistic conductor. Finally, by considering the correlation plots between

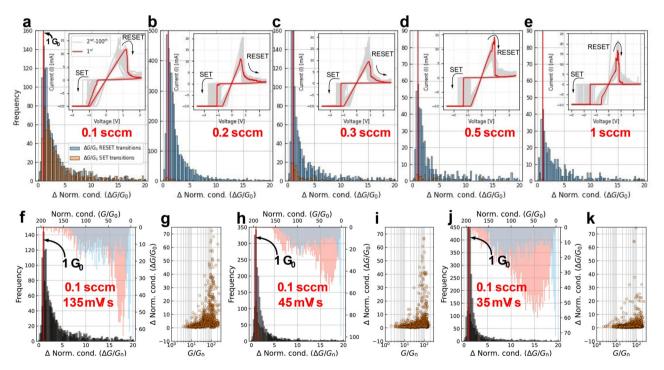


Figure 2. Histograms of the conductance transitions for each oxidation condition. (**a**) 0.1 sccm, (**b**) 0.2 sccm, (**c**) 0.3 sccm, (**d**) 0.5 sccm and (**e**) 1 sccm. The inset presents the overall aspect of the associated I–V loops. Note that very few transitions are found during the set event for oxidations larger than 0.1 sccm. This is due to the very abrupt set transition. Histograms of the conductance transitions for three different samples with the same oxidation condition but obtained with different ramp rates. (**f**) 135 mV/sec. (**h**) 45 mV/sec. (**j**) 35 mV/sec. The bottom x-axis indicates the histogram of the quantum transitions during the SET/RESET events and the upper x-axis indicates the histogram of conductance levels in units of G_0 found during the transition. The panels at the right of each histogram reports the correlation plot between the conductance level and the conductance transitions (**g**, **i**, **k**).

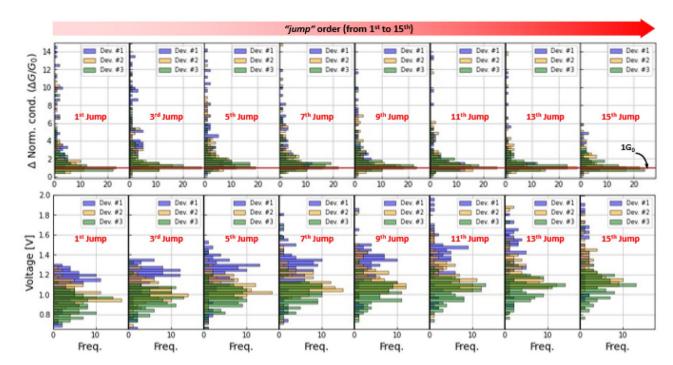


Figure 3. Histograms of the quantum conductance transitions as a function of the transition order (1st jump up to 15th jump) (first row). For three different samples, the transitions occur always around 1 G_0 . Interestingly, if the ramp rate is high, then the first jumps are roughly higher (4–5 G_0). Histograms of the quantum conductance transition voltages as a function of the transition order (1st jump up to 15th jump) (second row). As the ramp rate increases, the transitions occur at lower voltages. But for all cases, there is a shift to higher voltages as the order increases, which is expected.

normalized conductance levels at which the transitions take place (G/G_0) , and the transition itself $(\Delta G/G_0)$, it is possible to see in Fig. 2g,i,k (and in more detail in Supplementary Fig. 2) that the small transitions between quantized conductances take place mostly for conductance values close to G_0 . This behavior is reasonable since (i) the conductance state cannot reach lower conductance values unless a deep reset is performed (requiring higher voltages) with the consequent formation of a gap and (ii) conductance quantization is observable only when the lateral size of the constriction is comparable to the electron wavelength, which is not the case for highly conductive filaments¹³.

The different likelihood of the set and reset processes to exhibit conductance quantization effects as a function of the oxygen flow used for fabrication can be explained by considering the different mechanisms involved in each of these switching events³¹. In general, for oxide-based bipolar switching memristors, the set process is much faster and thus more abrupt than the reset process, which hinders the observation of quantization effects 12,32,33. During the set transition, the movement of ions is activated by both the electric field and Joule heating related to thermal effects, with Soret forces (thermophoresis) driving oxygen vacancies towards the conducting channel under a steep temperature gradient occurring in the oxygen-rich parent phase, therefore creating a positive feedback loop between field-induced and temperature effects. 34-37 On the other hand, during reset, the heatassisted and concentration gradient-driven Fick diffusion dominates over thermophoresis as the oxygen-poor conducting path is characterized by higher heat and electrical conductivity than the surrounding insulating matrix. This leads to the creation of a flatter temperature gradient resulting in a slower switching dynamic. It has been shown that in low conductive filaments (i.e. when approaching the HRS regime) the contribution of phonon heat conduction allows decreasing the temperature³⁶. This also means that the introduction of a higher degree of oxygen deficiency improves the electron and heat conduction in the functional layer, thus ultimately minimizing the contribution of thermal effects and making the field-driven drift of oxygen ions dominant, allowing for a higher controllability of the switching dynamics and the stabilization of intermediate resistance states both in the set and reset processes¹⁶.

To better understand the dynamics of the transitions, the trends depicted in Fig. 2 were further analyzed by considering the conductance transitions based on their order of occurrence. Here, we describe the transition order as the sequential number of the transition between two consecutive conductance levels, i.e. the first transition in the reset is of first order, the second jump is of second order, and so on. The histograms for the 1st, 2nd, 3rd, 5th, 7th, 9th, 11th, 13th, 15th conductance jumps (see top row in Fig. 3) are separately investigated for the three devices considered in Fig. 2f–k. As the transition order increases, the histograms of all three devices present a peak around ~ 1 G_0 . Notwithstanding this, the very first transitions show a different behavior, which might be attributed to the different ramp rate used for the voltage sweeps, as for the fastest ramp rate case, the first transitions are normally larger than the rest with a magnitude of ~ 4–5 G_0 . At the same time, and as expected, the average voltage (see second row in Fig. 3) for the occurrence of a given transition order shifts to higher voltages as the transition order increases. It is worth noticing that the average transition voltage increases with the

ramp rate of the voltage sweep, which is in agreement with previously reported experimental observations³⁸. These results are summarized in Fig. 4, where the solid lines indicate the median trend and the error bars the dispersion observed for each transition order. Note that the linear trend exhibited by the jump location-jump order curve in most of the investigated range seems to indicate that the filament rupture rate is approximately constant regardless of the actual size of the filament. Last but not least, the correlation plots between G/G_0 and $\Delta G/G_0$ for each transition order show a progressive evolution from the behavior observed for the 1st transition to that for the 15th transition (See supplementary Fig. 2). The 1st transitions are always observed at high conductance levels and for the samples measured with a voltage ramp rate of 135 mV/s, they show higher values (See supplementary Fig. 2, bottom row, left-most column), suggesting that these transitions are not transitions between quantized states. This is likely a morphology transition in which the consolidated filament becomes more granular and therefore more susceptible of closing independent leakage pathways. As the transition order increases, the magnitude of the transitions decreases to ~ 1 G_0 as the absolute conductance levels converge to G_0 as described in the previous section. This is ultimately consistent with a single ballistic conducting channel.

However, the trends for the slowest voltage ramp rates (35 and 45 mV/s, yellow and green markers in Fig. 3, top row and Supplementary Fig. 2, top and middle row,) to show conductance jumps occurring in multiples of G_0 units are remarkable as the initial conductance in the LRS reaches values up to 270 G_0 , a situation in which conduction cannot be associated with a nanosized ballistic channel. However, the change of conduction still occurs in G_0 units, and this is a striking finding that needs to be further elaborated on. A simplified physical picture of the phenomenon is depicted in Fig. 5, with a schematic representation of the conducting region in a highly conductive state and in a state where conduction converges to 1 G_0 close to the end of the reset process (Fig. 5a). According to this phenomenological interpretation, the conducting region consists of multiple interacting channels, each with a conductance of about 1 G_0 and in close proximity of each other, forming a robust conducting path in the LRS. Then, during reset, this path gets narrower and narrower by the elimination of single channels, inducing a conductance change equal to the number of channels annihilated (see Fig. 5b). Therefore, even though the conduction is apparently diffusive because of the contribution of several channels, in the long run the change of conductance shows quantization effects. Of course, since the conductor in this case is formed from a metal-rich phase in the oxide, i.e., oxygen vacancies, dispersion should be expected around the quantized levels, as it is observed in the experimental data. Since the reset process consists of the elimination of single channels, a transition is expected from diffusive transport to the mesoscopic, ballistic regime below a critical number of remaining channels, and then to tunnelling with the closure of all channels. We expect that the above-described phenomenon can be universally expanded to other oxide-based RRAM devices that show quasi-analog/semi-gradual switching characteristics, including the well-known Ta-O and Hf-O systems^{27,31,39}. Also, the fact that the majority of transitions occurs with a magnitude of $\sim 1~G_0$ independent of the yttria layer's oxygen content suggests that the use of a scavenger layer should also yield similar results.

Summary

In summary, the quasi-analog nature of the reset process was analyzed in oxygen engineered yttrium oxide-based RRAM devices with special emphasis on the conductance transitions between intermediate resistive states. Remarkably, these transitions are found to occur predominantly with a magnitude equal to the conductance quantum (G_0) even at resistance levels far below 12.9 k Ω , i.e., where electron transport is expected to be diffusive. This finding suggests that the changes in conductance are quantized, even if conduction itself is apparently not.

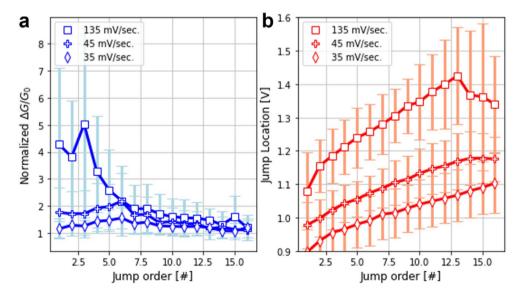
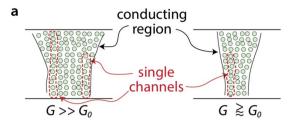


Figure 4. (a) Median values for the magnitude of the transitions and for the (b) voltage at which each jump occurred. The trends regarding the jump order and the ramp rate are evident. The higher the ramp rate the lower value of the normalized conductance jumps for the low order jumps.



bConductance jumps: elimination of single channels

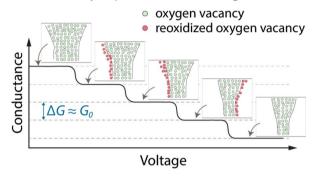


Figure 5. Schematic representation of the conducting region and its morphological changes occurring during reset. (a) conducting path morphology in a high and a low conducting state. The conducting region is comprised of several neighboring monomode conducting channels that form a diffusive and a ballistic transport path in the high and low conducting states, respectively. (b) Conductance jumps occurring in the reset in multiples of G_0 due to elimination of single conduction channels within the conducting region. The conductance becomes quantized when the lateral size of filament is shrunk to a few atoms width.

As part of this process and with the increment of the transition order, the voltage at which the conductance jumps occur shifts to higher values and a clear convergence of the jump magnitude towards G_0 is observed. The associated trends are independent of the ramp rates investigated in this study. To provide an explanation to the observed phenomenon, a plausible interpretation of the conducting filament is presented, wherein the conducting region is formed by multiple monomode ballistic conductors in close vicinity to each other. The transition between two adjacent intermediate conductance levels occurs through the elimination of single channels, therefore inducing changes in multiples of G_0 . These findings suggest that besides the study of absolute conductance quantization in filamentary-type RRAM devices, the change of conductance should also be investigated as a possible tool for applications, like processing-in-memory and neuromorphic computing architectures, where multilevel data storage and analog switching characteristics are required.

Data availability

The datasets generated during and/or analysed during the current study are available from the corresponding author on reasonable request.

Received: 27 June 2023; Accepted: 13 December 2023 Published online: 11 January 2024

References

- 1. Zhang, Z., Wu, Y., Philip Wong, H. S. & Simon Wong, S. Nanometer-scale HfOx RRAM. IEEE Electron. Device Lett. 34, 1005–1007 (2013).
- 2. Torrezan, A. C., Strachan, J. P., Medeiros-Ribeiro, G. & Williams, R. S. Sub-nanosecond switching of a tantalum oxide memristor. *Nanotechnology* 22, 485203 (2011).
- 3. Wouters, D. J., Waser, R. & Wuttig, M. Phase-change and redox-based resistive switching memories. *Proc. IEEE* 103, 1274–1288 (2015).
- 4. Kumar, S. et al. Oxygen migration during resistance switching and failure of hafnium oxide memristors. Appl. Phys. Lett. 110, 103503 (2017).
 5. Petzold, S. et al. Heavy ion radiation effects on hafnium oxide-based resistive random access memory. IEEE Trans. Nucl. Sci. 66,
- 1715–1718 (2019).

 6. Vogel, T. *et al.* Structural and electrical response of emerging memories exposed to heavy ion radiation. *ACS Nano* **16**, 14463–14478
- 6. Vogel, 1. et al. Structural and electrical response of emerging memories exposed to heavy ion radiation. ACS Nano 16, 14463–144/8 (2022).
- Waser, R., Dittmann, R., Staikov, C. & Szot, K. Redox-based resistive switching memories nanoionic mechanisms, prospects, and challenges. Adv. Mater. 21, 2632–2663. https://doi.org/10.1002/adma.200900375 (2009).
- Sharath, S. U. et al. Impact of oxygen stoichiometry on electroforming and multiple switching modes in TiN/TaOx/Pt based ReRAM. Appl. Phys. Lett. 109, 173503 (2016).

- 9. Petzold, S. et al. Gradual reset and set characteristics in yttrium oxide based resistive random access memory. Semicond. Sci. Technol. 34, 075008 (2019).
- 10. Datta, S. Electronic Transport in Mesoscopic Systems (Cambridge University Press, 1997).
- 11. Xue, W. et al. Recent advances of quantum conductance in memristors. Adv. Electron. Mater. 5, 1800854. https://doi.org/10.1002/aelm.201800854 (2019).
- 12. Li, Y. et al. Conductance quantization in resistive random access memory. Nanoscale Res. Lett. 10, 1-30 (2015).
- Milano, G. et al. Quantum conductance in memristive devices: Fundamentals, developments, and applications. Adv. Mater. 34, 2201248. https://doi.org/10.1002/adma.202201248 (2022).
- Pan, F., Gao, S., Chen, C., Song, C. & Zeng, F. Recent progress in resistive random access memories: Materials, switching mechanisms, and performance. Mater. Sci. Eng. R Rep. 83, 1–59. https://doi.org/10.1016/j.mser.2014.06.002 (2014).
- Zhu, X. et al. Observation of conductance quantization in oxide-based resistive switching memory. Adv. Mater. 24, 3941–3946 (2012).
- 16. Petzold, S. et al. Tailoring the switching dynamics in yttrium oxide-based RRAM devices by oxygen engineering: From digital to multi-level quantization toward analog switching. Adv. Electron. Mater. 6, 1–13 (2020).
- Piros, E. et al. Role of oxygen defects in conductive-filament formation in Y₂O₃-based analog RRAM devices as revealed by fluctuation spectroscopy. Phys. Rev. Appl. 14, 034029 (2020).
- Piros, E. et al. Enhanced thermal stability of yttrium oxide-based RRAM devices with inhomogeneous Schottky-barrier. Appl. Phys. Lett. 117, 013504 (2020).
- 19. Aguirre, F. L. et al. Fast fitting of the dynamic memdiode model to the conduction characteristics of RRAM devices using convolutional neural networks. *Micromachines* 13, 2002 (2022).
- Rushchanskii, K. Z., Blügel, S. & Ležaić, M. Ab initio phase diagrams of Hf-O, Zr-O and Y-O: A comparative study. Faraday Discuss. 213, 321-337 (2019).
- 21. Onisawa, K. I. et al. Dielectric properties of rf-sputtered Y₂O₃ thin films. J. Appl. Phys. 68, 719 (1990).
- 22. Zaumseil, P. RCRefSim (Rocking curve and reflectivity simulation). IHP Frankf. Oder (2005).
- 23. Mehonic, A. et al. Quantum conductance in silicon oxide resistive memory devices. Sci. Rep. 31(3), 1-8 (2013).
- 24. Miranda, E. et al. Nonlinear conductance quantization effects in CeOx/SiO2-based resistive switching devices. Appl. Phys. Lett. 101, 012910 (2012).
- 25. Chen, C. et al. Conductance quantization in oxygen-anion-migration-based resistive switching memory devices. *Appl. Phys. Lett.* **103**, 043510 (2013).
- Tappertzhofen, S., Valov, I. & Waser, R. Quantum conductance and switching kinetics of AgI-based microcrossbar cells. Nano-technology 23, 145703 (2012).
- 27. Syu, Y. E. et al. Atomic-level quantized reaction of HfOx memristor. Appl. Phys. Lett. 102, 172903 (2013).
- 28. Tukey, J. W. Exploratory Data Analysis (Addison-Wesley Publishing Company, 1977).
- 29. Miranda, E., Mehonic, A., Ng, W. H. & Kenyon, A. J. Simulation of cycle-to-cycle instabilities in SiO _x -based ReRAM devices using a self-correlated process with long-term variation. *IEEE Electron. Device Lett.* **40**, 28–31 (2019).
- 30. Krishnan, K., Muruganathan, M., Tsuruoka, T., Mizuta, H. & Aono, M. Quantized conductance operation near a single-atom point contact in a polymer-based atomic switch. *Jpn. J. Appl. Phys.* **56**, 06GF02 (2017).
- 31. Sharath, S. U. et al. Control of switching modes and conductance quantization in oxygen engineered HfOx based memristive devices. Adv. Funct. Mater. 27, 1700432 (2017).
- 32. Lee, Y., Park, J., Chung, D., Lee, K. & Kim, S. Multi-level cells and quantized conductance characteristics of Al2O3-based RRAM device for neuromorphic system. *Nanoscale Res. Lett.* 17, 84 (2022).
- 33. Long, S. et al. Quantum-size effects in hafnium-oxide resistive switching. Appl. Phys. Lett. 102, 183505 (2013).
- Strukov, D. B., Alibart, F. & Stanley Williams, R. Thermophoresis/diffusion as a plausible mechanism for unipolar resistive switching in metal-oxide-metal memristors. Appl. Phys. Mater. Sci. Process. 107, 509–518 (2012).
- 35. Yang, J. J., Strukov, D. B. & Stewart, D. R. Memristive devices for computing. Nat. Nanotechnol. 8, 13–24. https://doi.org/10.1038/nnano.2012.240 (2013).
- 36. Ielmini, D. Modeling the universal set/reset characteristics of bipolar RRAM by field- and temperature-driven filament growth. *IEEE Trans. Electron Devices* **58**, 4309–4317 (2011).
- 37. Miao, F. *et al.* Anatomy of a nanoscale conduction channel reveals the mechanism of a high-performance memristor. *Adv. Mater.* **23.** 5633–5640 (2011).
- 38. Maestro-Izquierdo, M., Gonzalez, M. B., Campabadal, F., Sune, J. & Miranda, E. A new perspective towards the understanding of the frequency-dependent behavior of memristive devices. *IEEE Electron Device Lett.* 42, 565–568 (2021).
- 39. Yi, W. et al. Quantized conductance coincides with state instability and excess noise in tantalum oxide memristors. Nat. Commun. 7, 11142 (2016).

Acknowledgements

F. Aguirre and E. Piros contributed equally to this work. This work was conducted in the frame of the project StorAlge. The project has received funding from the ECSEL Joint Undertaking (JU) under grant agreement No 101007321. The JU receives support from the European Union's Horizon 2020 research and innovation programme and France, Belgium, Czech Republic, Germany, Italy, Sweden, Switzerland, Turkey. Funding by the Federal Ministry of Education and Research (BMBF) under contract 16MEE0154 is gratefully acknowledged. Funding was received within the ECSEL Joint Undertaking project WAKeMeUP in collaboration with the European Union's H2020 Framework Program (H2020/2014-2020) and National Authorities, under grant agreement no. 783176. Funding by the Federal Ministry of Education and Research (BMBF) under contract 16ESE0298 is gratefully acknowledged. Also, funding from DFG under project AL 560/13-2, and AL 560/21-1 are gratefully acknowledged. The authors also acknowledge the financial support provided by Merck KGaA, Darmstadt, Germany, in the framework of the joint research platform by the company and TU Darmstadt called "Sustainability Hub". E.M. and J.S. acknowledge the support provided by the European project MEMQuD, code 20FUN06, which has received funding from the EMPIR programme co-financed by the Participating States and from the European Union's Horizon 2020 research and innovation programme. E.M and J.S. also acknowledge project PID2022-139586NB-C41 from the Ministerio de Ciencia e Innovación, Spain. F.A. acknowledges financial support from MICINN (Spain) through the programme Juan de la Cierva-Formación grant number FJC2021-046808-I.

Author contributions

Conceptualization, F.L.A., E.P., E.M. and L.A.; methodology, F.L.A., E.P., E.M. and L.A.; software, F.L.A. and E.M.; validation, F.L.A., E.P., E.M. and L.A.; formal analysis, F.L.A., E.P., E.M. and L.A.; investigation, F.L.A., E.P., L.A.,

and E.M.; resources, C.H., J.S., E.M and L.A.; data curation, E.P. and F.L.A.; writing—original draft preparation, E.P., F.L.A., E.M. and L.A.; writing—review and editing, E.P., F.L.A., N.K., T.V., S.P., J.G., T.O., K.H., C.H., J.S., E.M. and L.A.; visualization, E.P., F.L.A., E.M. and L.A.; supervision, L.A. and E.M.; project administration, C.H., J.S., E.M. and L.A.; funding acquisition, C.H., J.S., E.M. and L.A. All authors have read and agreed to the published version of the manuscript.

Funding

Open Access funding enabled and organized by Projekt DEAL.

Competing interests

The authors declare no competing interests.

Additional information

Supplementary Information The online version contains supplementary material available at https://doi.org/10.1038/s41598-023-49924-2.

Correspondence and requests for materials should be addressed to F.L.A. or E.P.

Reprints and permissions information is available at www.nature.com/reprints.

Publisher's note Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.

Open Access This article is licensed under a Creative Commons Attribution 4.0 International License, which permits use, sharing, adaptation, distribution and reproduction in any medium or format, as long as you give appropriate credit to the original author(s) and the source, provide a link to the Creative Commons licence, and indicate if changes were made. The images or other third party material in this article are included in the article's Creative Commons licence, unless indicated otherwise in a credit line to the material. If material is not included in the article's Creative Commons licence and your intended use is not permitted by statutory regulation or exceeds the permitted use, you will need to obtain permission directly from the copyright holder. To view a copy of this licence, visit http://creativecommons.org/licenses/by/4.0/.

© The Author(s) 2024